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FEFF8

The FEFF Project  
Department of Physics  
University of Washington

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# Abstract

*Ab initio* self-consistent real space multiple-scattering code for simultaneous calculations of x-ray-absorption spectra and electronic structure. The approach builds in core-hole effects and can include local fields (TDLDA). Output includes extended x-ray-absorption fine structure (EXAFS), full multiple scattering calculations of various x-ray absorption spectra (XAS) and projected local densities of states (LDOS). The spectra include x-ray absorption near edge structure (XANES), x-ray natural and magnetic circular dichroism (XNCD and XMCD), spin polarized x-ray absorption spectra (SPXAS and SPEXAFS), and non-resonant x-ray emission spectra (XES). Calculation of the x-ray scattering amplitude (Thomson and anomalous parts) is also possible.

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This document can also be found in wiki format on the web at:

<http://leonardo.phys.washington.edu/feff/wiki>

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# Chapter 1

## Synopsis

FEFF8 calculates extended x-ray-absorption fine structure (EXAFS), x-ray-absorption near-edge structure (XANES), x-ray natural circular dichroism (XNCD), spin-dependent calculations of x-ray magnetic dichroism (XMCD) and spin polarized x-ray absorption (SPXAS and SPEXAFS), nonresonant x-ray emission (XES), electronic structure including local densities of states (LDOS), and x-ray elastic scattering amplitude  $f = f_0 + f' + if''$ . FEFF uses an *ab initio* self-consistent real space multiple scattering (RSMS) approach for clusters of atoms ( $Z < 99$ ), including polarization dependence, core-hole effects, and local field corrections. Calculations are based on an all-electron, real space relativistic Green's function formalism with no symmetry requirements. The method combines both full multiple scattering based on LU or Lanczos algorithms and a high-order path expansion based on the Rehr-Albers multiple scattering formalism. For a quick start or self-guided tutorial we suggest that new users try a few of the examples in Section 4. For details on use of the code, examples and suggestions for calculation strategies, see Sections 2, 3, and 4. For details about the algorithms used see the discussion for the appropriate module in Section 2 and, for more detail, the published references listed in Appendix C. The principal investigator of the FEFF project is:

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FEFF is written in ANSI Fortran 77, with the non-standard extension of double precision complex (`complex*16`) variables. It requires at least 18 megabytes (MB) of available memory (RAM) to run. For XANES calculations, one generally needs more memory (about 60 MB of RAM for a cluster of 100 atoms, about 170 MB for a cluster of 200 atoms, and so on). See Appendix B for installation instructions.

Please contact the authors concerning any problems with the code. See Appendix F for trouble-shooting hints and problem/bug reports or the FAQ on the FEFF WWW pages (see below).

The full FEFF8 code is copyright protected software and users must obtain a license from

the University of Washington Office of Technology Transfer for its use. See Appendix A for complete copyright notice and other details. Documentation and information on how to obtain the code can be found at the FEFF Project WWW URL:

<http://leonardo.phys.washington.edu/feff/>

or by e-mail to the FEFF Project Coordinator at [feff@phys.washington.edu](mailto:feff@phys.washington.edu).

Please cite FEFF and an appropriate FEFF reference if the code or its results are used in published work. See Appendix C for a list of appropriate citations. The main published reference for FEFF8 is: *Real Space Multiple Scattering Calculation and Interpretation of X-ray Absorption Near Edge Structure*, A.L. Ankudinov, B. Ravel, J.J. Rehr, and S.D. Conradson Phys. Rev. B**58**, 7565 (1998), and a manuscript dealing the improvements added in FEFF8.20 is A. L. Ankudinov, C. E. Bouldin, J. J. Rehr, J. Sims, and H. Hung, Phys. Rev. B**65**, 104107 (2002).

**FEFF Project Developers** – Several developers contributed to the FEFF8.00 code, and its advanced versions FEFF8.10, FEFF8.20, and FEFF8.40.

Alexei Ankudinov (ALA) is the principal developer of FEFF8 series. ALA implemented the automated self-consistent potential algorithm, and added calculations of LDOS, the Fermi level, and charge transfer. Bruce Ravel is the principal developer of the full multiple scattering (FMS) algorithm in FEFF8, which uses LU decomposition and is necessary for the SCF loop, DOS and for FMS/LU XANES calculations. For version FEFF8.10 ALA added calculations of elastic scattering amplitude, x-ray natural circular dichroism and nonresonant x-ray emission. ALA is largely responsible for the new version FEFF8.20, which has an improved input/output structure to facilitate interaction with other codes. He also added quadrupolar transitions and implemented fast Lanczos methods for the iterative FMS algorithm. In collaboration with A. Nesvizhskii, new routines for sum-rules applications have been added. Anna Poiarkova and Patrick Konrad contributed new codes for calculating multiple-scattering Debye–Waller factors and anharmonic contributions. Matthew Newville added an improved padded ascii output structure to facilitate the interface to XAFS analysis codes and portability between different machines. Jim Sims (NIST) modified the code for MPI based parallel execution, in collaboration with C. Bouldin and JJR.

The high-order multiple-scattering routines, pathfinder, and input/output routines were largely developed by Steven Zabinsky and JJR for FEFF5 and 6, and are still in use. The Hedin–Lundqvist self-energy and phase shift routines were developed for early versions of FEFF in collaboration with Jose Mustre de Leon, Dan Lu and R.C. Albers and are still part of FEFF8.

The authors thank many users of experimental versions of the FEFF8 code for feedback, suggestions and bug reports. The authors also thank R.C. Albers, K. Baberschke, C. Bouldin, C. Brouder, G. Brown, S.D. Conradson, F. Farges, G. Hug, M. Jaouen, J. Sims, and E. Stern for useful comments.

| font                           | denotes                               |
|--------------------------------|---------------------------------------|
| SMALL CAPS                     | names of programs                     |
| typewriter font                | contents of files                     |
| 'quoted typewriter font'       | file names                            |
| ROMAN CAPITALS                 | names of cards in the 'feff.inp' file |
| <i>slanted typewriter font</i> | commands executed at a command line   |

Table 1.1: Typographic conventions in this document

## Chapter 2

# Input File Control Cards

The main program FEFF reads single file (`'feff.inp'`) created by the user and executes the various program modules described below. An auxiliary FORTRAN program (ATOMS), developed by Bruce Ravel, is provided which can generate the `'feff.inp'` file from crystallographic input parameters. A perl-based graphical user interface to ATOMS is also available. Information about ATOMS can be found on the WWW at

<http://cars9.uchicago.edu/~ravel/software/aboutatoms.html>

This section describes `'feff.inp'` and the commands that tell FEFF what to do. It may be helpful to look at one or more of the sample input files in Section 4 while reading this section. The input file for FEFF8 is similar to FEFF version 5 through 7, except that several additional options have been added to permit self-consistent potential generation, full multiple scattering XANES calculations, polarization dependence, and others listed in Section 2.1. However FEFF8 is backwards compatible and supports earlier FEFF5-7 style input files.

The input file `'feff.inp'` is a loosely formatted, line oriented text file. Each type of input read by the program is on a line which starts with a *card* or *keyword* and, in some cases, is followed by alpha-numeric data. All card arguments listed below inside square brackets are optional. The sequence of keyword cards is arbitrary. If any card or optional data is omitted, default values are used; an exception is the POTENTIALS card, which is always required. Alpha-numeric values are listed in free format, separated by blanks. Tab characters are *not allowed* (due to Fortran 77 portability constraints) and may cause confusing error messages. Any characters appearing after the card and any required or optional data on a given line are ignored by FEFF and can be used as end-of-line comments. All distances are in Ångstroms and energies in eV. Spaces between lines (empty lines) are ignored. Any line beginning with an asterisk (\*) is regarded as a comment and is also ignored.

After reading the `'feff.inp'` file, the calculations of various spectroscopies are done sequentially in six steps:

1. The scattering potentials are calculated using atomic overlap (Mattheiss) prescription or self-consistently using an automated SCF loop. Absolute energies are estimated. (module **POT**, potentials).

2. The scattering phase shifts, dipole matrix elements, x-ray cross-section and angular momentum projected density of states (LDOS) are calculated. (module **XSPH**, cross-section and phases).
3. Full multiple scattering XANES calculations are done for a specified cluster size. (module **FMS**, full multiple scattering).
4. The leading multiple scattering paths for the cluster are enumerated. (module **PATHS**).
5. The effective scattering amplitudes  $f_{\text{eff}}$  and other XAFS parameters are calculated for each scattering path. (module **GENFMT**, general-path F-matrix calculation).
6. The XAFS parameters from one or more paths are combined to calculate a total XAFS or XANES spectrum. (module **FF2CHI**, scattering amplitude to chi).

This section describes how to control each module using the ‘**feff.inp**’ file.

## 2.1 Complete list of FEF8 control cards

The list of ‘**feff.inp**’ options falls into three categories, *standard* options frequently and easily used, *useful* options that are often used, and *advanced* options that are seldom necessary but may be helpful in some cases. Every card in ‘**feff.inp**’ file will influence the calculations, starting from some module. Thus for better understanding how each module can be affected by the input cards, we list three categories (standard, useful, and advanced) for each module separately.

### Module 0 ..... RDINP

**Purpose of Module:** Read input data

**Standard cards:** ATOMS, CONTROL, PRINT, and TITLE

**Useful Cards:** END, and RMULTIPLIER

**Advanced Cards:** CFAVERAGE and OVERLAP

### Module 1 ..... POT

**Purpose of Module:** Calculate (self-consistent) scattering potentials and Fermi energy

**Standard cards:** POTENTIALS, AFOLP, S02

**Useful Cards:** EXCHANGE, NOHOLE, RGRID, SCF, and UNFREEZE

**Advanced Cards:** FOLP, INTERSTITIAL, ION, and SPIN

### Module 2 ..... XSPH

**Purpose of Module:** Calculate cross-section and phase shifts and  $\ell$ DOS

**Standard cards:** EXAFS, XANES, EDGE and HOLE,

---

|                                                                                     |        |
|-------------------------------------------------------------------------------------|--------|
| <b>Useful Cards:</b> ELLIPTICITY, LDOS, MULTIPOLE, POLARIZATION, and TDLDA          |        |
| <b>Advanced Cards:</b> DANES, FPRIME, RPHASES, XES, and XNCD or XMCD                |        |
| <b>Module 3</b> .....                                                               | FMS    |
| <b>Purpose of Module:</b> Calculate full multiple scattering for XAS                |        |
| <b>Standard cards:</b> FMS                                                          |        |
| <b>Useful Cards:</b> DEBYE                                                          |        |
| <b>Advanced Cards:</b>                                                              |        |
| <b>Module 4</b> .....                                                               | PATHS  |
| <b>Purpose of Module:</b> Path enumeration                                          |        |
| <b>Standard cards:</b> RPATH                                                        |        |
| <b>Useful Cards:</b> NLEG                                                           |        |
| <b>Advanced Cards:</b> PCRITERIA and SS                                             |        |
| <b>Module 5</b> .....                                                               | GENFMT |
| <b>Purpose of Module:</b> Calculate scattering amplitudes and other XAFS parameters |        |
| <b>Standard cards:</b>                                                              |        |
| <b>Useful Cards:</b> CRITERIA                                                       |        |
| <b>Advanced Cards:</b> IORDER and NSTAR                                             |        |
| <b>Module 6</b> .....                                                               | FF2CHI |
| <b>Purpose of Module:</b> Calculate final output.                                   |        |
| <b>Standard cards:</b> DEBYE                                                        |        |
| <b>Useful Cards:</b> CORRECTIONS and SIG2                                           |        |
| <b>Advanced Cards:</b>                                                              |        |

These data types are listed below in the same order as in the table above. Each description is of this form:

◆ **CARD**    argument list *type*  
 The type is one of *Standard*, *Useful*, or *Advanced*. The argument list is a brief statement of the valid arguments to the card. The text description will explain the arguments and their uses more fully. Example uses of the card look like this:

\* brief description of the example  
 CARD    argument list

## 2.2 Main Control Cards

These cards in this section are not associated with any particular module, but are used throughout the FEFF calculation. The ATOMS card is used to specify the absorbing atom and its environment. (If atomic coordinates are not known, then the OVERLAP card can be used to construct approximate potentials). Without this structural information no calculations can be done. The CONTROL card is used to selectively run parts of FEFF. The PRINT card controls which output files are written by the modules.

### ◆ ATOMS

*Standard*

Cartesian coordinates (in Ångstroms) and unique potential indices of each atom in the cluster are entered following the ATOMS card, one per line. See the discussion of the POTENTIALS card. An auxiliary code, ATOMS written by Bruce Ravel, is supplied with FEFF to generate the ATOMS list from crystallographic data. See the document file for ATOMS for more information.

```

ATOMS
* x      y      z      ipot      SF6 molecule
  0.0    0.0    0.0      0      S K-shell hole

  1.56   0.00   0.00     1      F 1st shell atoms
  0.00   1.56   0.00     1
  0.00   0.00   1.56     1
 -1.56   0.00   0.00     1
  0.00  -1.56   0.00     1
  0.00   0.00  -1.56     1

```

### ◆ CONTROL ipot ixsph ifms ipaths igenfmt iff2chi

*Standard*

The CONTROL card lets you run one or more of the modules separately. There is a switch for each module: 0 means not to run that module, 1 meaning to run it. You can do the whole run in sequence, one module at a time, but you *must* run all modules sequentially. *Do not* skip modules: CONTROL 1 1 1 0 0 1 is incorrect. The default is CONTROL 1 1 1 1 1 1, i.e. run all 6 modules.

```

* example 1
* calculate self consistent potentials, phase shifts and fms only
CONTROL 1 1 1 0 0 0 ipot ixsph ifms ipaths igenfmt iff2chi

* example 2
* run paths, genfmt and ff2chi; do not run pot, xsph, fms
* this run assumes previous modules have already been run and
* adds MS paths between rfms and rpath to the MS expansion
CONTROL 0 0 0 1 1 1 ipot ixsph ifms ipaths igenfmt iff2chi

```

◆ **PRINT** ppot pxsph pfms ppaths pgenfmt pff2chi *Standard*

The PRINT card determines which output files are printed by each module. The default is print level 0 for each module. See Section 3 for details about the contents of these files.

```
* add crit.dat and feffNNNN.dat files to minimum output
PRINT 0 0 0 1 0 3
```

The print levels for each module are summarized in Table 2.1 on page 9.

◆ **TITLE** title\_line *Standard*

User supplied title lines. You may have up to 10 of these. Titles may have up to 75 characters. Leading blanks in the titles will be removed.

```
TITLE Andradite (Novak and Gibbs, Am.Mineral 56,791 1971)
TITLE K-shell 300K
```

◆ **END** *Useful*

The END card marks the end of portion of the ‘feff.inp’ file read by FEFF. All data following the END card is ignored. Without an END card, the entire input file is read.

```
END ignore any lines in feff.inp that follow this card
```

◆ **RMULTIPLIER** rmult *Useful*

With RMULTIPLIER all atomic coordinates are multiplied by the supplied value. This is useful to adjust lattice spacing, for example, when fractional unit cell coordinates are used. By default, rmult=1.

```
*increase distances by 1%
RMULTIPLIER 1.01
```

◆ **CFAVERAGE** iphabs nabs rclabs *Advanced*

A “configuration” average over absorbers is done if the CFAVERAGE card is given. CFAVERAGE currently assumes phase transferability, which is usually good for EXAFS calculations, but may not be accurate for XANES. Note that the CFAVERAGE card is currently incompatible with the DEBYE card for options other than correlated Debye model (`idwopt` > 0).

**iphabs** potential index for the type of absorber over which to make the configuration average (potential 0 is also allowed)

**nabs**

the configuration average is made over the first **nabs** absorbers in the ‘feff.inp’ file

| module | print levels                                                                                                                                                                                                                                                                                                                                                                                          |
|--------|-------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------|
| pot    | 0 write 'pot.bin' only<br>1 add 'misc.dat'<br>2 add 'potNN.dat'<br>5 add 'atomNN.dat'                                                                                                                                                                                                                                                                                                                 |
| xsph   | 0 write 'phase.bin'; with LDOS card add 'ldosNN.dat' and 'logdos.dat'<br>1 add 'axafs.dat'<br>2 add 'phaseNN.dat' and 'phminNN.dat'<br>3 add 'ratio.dat' for XMCD normalization and 'emesh.dat'                                                                                                                                                                                                       |
| fms    | 0 write 'fms.bin'                                                                                                                                                                                                                                                                                                                                                                                     |
| paths  | 0 write 'paths.dat' only<br>1 add 'crit.dat'<br>3 add 'fbeta' files (plane wave $ f(\beta) $ approximations)<br>5 Write only 'crit.dat' and do not write 'paths.dat'. (This is useful when exploring the importance of paths for large runs.)                                                                                                                                                         |
| genfmt | 0 write 'files.dat' and 'list.dat', and 'feff.dat's with paths with importance greater than or equal to two thirds of the curved wave importance criterion<br>1 keep all paths written to 'feff.dat's                                                                                                                                                                                                 |
| ff2chi | 0 write 'chi.dat' and 'xmu.dat'<br>1 add 'sig2.dat' with Debye–Waller factors;<br>2 add 'chipNNNN.dat' ( $\chi(k)$ from each path individually) This can quickly fill up your disk if you're doing a large run.<br>3 add 'feffNNNN.dat' (input files for Matt Newville's FEFFIT program) and 'files.dat', and do not add 'chipNNNN.dat' files.<br>4 add both 'feffNNNN.dat' and 'chipNNNN.dat' files. |

Table 2.1: Print levels controlling output files from the modules.

of type `iphabs`. You don't have to have potential of index 0 in your input file when using the `CFAVERAGE` card, but you must have the same type of potential for `iph=0` and `iph=iphabs`. The configurational average is done over ALL atoms of type `iphabs`, if `nabs` is less or equal zero.

#### `rclabs`

radius to make small atom list from a bigger one allowed in '`feff.inp`'. Currently the parameter controlling the maximum size of the list, `natxx`, is 100,000 but can be increased. The pathfinder will choke on too big an atoms list. You must choose `rclabs` to have less than 1,000 atoms in small atom list. If your cluster is less 1000 atoms simply use `rclabs=0` or negative always to include all atoms.

Default values are `iphabs=0`, `nabs=1`, `rclabs=0` (where `rclabs = 0` means to consider an infinite cluster size).

```
*average over all atoms with iph=2 in feff.inp with less than 1000 atoms
CFAVERAGE 2 0 0
```

#### ◆ OVERLAP iph

*Advanced*

The OVERLAP card can be used to construct approximate overlapped atom potentials when atomic coordinates are not known or specified. If the atomic positions are listed following the ATOMS cards, the OVERLAP cards are not needed. FEFF8 will stop if both ATOMS and OVERLAP cards are used. The OVERLAP card contains the potential index of the atom being overlapped and is followed by a list specifying the potential index, number of atoms of a given type to be overlapped and their distance to the atom being overlapped. The examples below demonstrate the use of an OVERLAP list. This option can be useful for initial single scattering XAFS calculations in complex materials where very little is known about the structure.

You should verify that the coordination chemistry built in using the OVERLAP cards is realistic. It is particularly important to specify all the nearest neighbors of a typical atom in the shell to be overlapped. The most important factor in determining the scattering amplitudes is the atomic number of the scatterer, but the coordination chemistry should be approximately correct to ensure good scattering potentials. Thus it is important to specify as accurately as possible the coordination environment of the scatterer. Note: If you use the OVERLAP card then you cannot use the FMS or SCF cards. Also the pathfinder won't be called and you must explicitly specify single scattering paths using SS card, which is described in Section 2.6.

\* Example 1. Simple usage

\* Determine approximate overlap for central and 1st nn in Cu

```
OVERLAP 0          determine overlap for central atom of Cu
  *iphovr  novr  rovr      * ipot, number in shell, distance
    1      12    2.55266
```

```
OVERLAP 1          determine approximate overlap for 1st shell atoms
  *iphovr  novr  rovr      * ipot, number in shell, distance
```

```

0          12      2.55266

* Example 2. More precise usage
* Determine approximate overlap for 3rd shell atoms of Cu
OVERLAP 3
0  1 2.55266      ipot, number in shell, distance
1  4 2.55266
2  7 2.55266
2  6 3.61000
2 24 4.42133

```

## 2.3 POT: Scattering potentials

Initially the free atom potentials of each atomic type are calculated as if the atoms were isolated in space using a relativistic Dirac–Fock atom code. Scattering potentials are calculated by overlapping the free atom densities within the muffin tin approximation (Mattheiss prescription), and then including the Hedin–Lundqvist/Quinn self energy for excited states. Non overlapping muffin-tin radii are determined automatically from the calculated Norman radii. Automatic overlapping of muffin tin spheres (see the AFOLP card) is done by default, since it typically leads to better results than non overlapping muffin-tin spheres. FEFF8 can also calculate self-consistent potentials by successively calculating the electron density of states, electron density and Fermi level at each stage within a small cluster and then iterating, using the Mattheiss prescription for the initial iteration.

XAFS spectra are referenced to the threshold Fermi level. This quantity is best determined with the self-consistent field procedure (typically to within a fraction of an eV), or (less accurately but faster) can be estimated from the electron gas result at the mean interstitial density within Mattheiss prescription, as in FEFF7. An absolute energy scale is obtained by an atomic calculation of the total energy of the system with and without the core-hole. Atomic configurations and core-hole lifetimes are built in, and mean free paths are determined from the imaginary part of the average interstitial potential, including self-energy and lifetime contributions.

The potential calculations need as input only the atomic number of the atoms, and, for the absorbing atom, the type of the core hole being considered. To do the overlapping of the unique potentials, the neighboring atoms must be identified, either by position (from a list of the Cartesian coordinates of each atom) or by explicit overlapping instructions using the OVERLAP card described in Section 2.3.

To save time the code calculates the overlapped atom potential for each unique potential only once, using as a sample geometry for an atom with a given unique potential index that for the atom that is closest to the absorbing atom. Thus it is essential that the neighborhood of each sample atom be appropriate.

This automatically overlaps all muffin-tins to a specified maximum value (default `folpx=1.15`) to reduce the effects of potential discontinuities at the muffin-tins. Automatic overlapping is done by default and is useful in highly inhomogeneous materials. Typical values of the overlapping fraction should be between 1.0 and 1.3. See FOLP for a non-automated version. Automatic overlapping is done by default; to switch overlapping off, use 1.0 as the argument for AFOLP.

```
* touching muffin-tins; do not use automatic overlapping
AFOLP 1.0
```

#### ◆ EDGE label s02

*Standard*

The EDGE card is equivalent to the HOLE card, but you don't have to look up the appropriate integer index. Simply use the hole label: N0 means no hole, K means  $K$ -shell, L1 means  $L_I$ , and so on. Calculations with very shallow edges, e.g.  $M$ -shells and higher are not well tested; please complain to the authors if you encounter problems. As with the HOLE card you may also use the integer index instead of the label. All comments for HOLE card are valid for EDGE card - see the description below. Thus if the entry for  $S_0^2$  is less than 0.1,  $S_0^2$  will be estimated from atomic overlap integrals.

```
* L1-shell core hole, S02 = 1
EDGE L1 1.0
```

#### ◆ HOLE ihole s02

*Standard*

The HOLE card includes the hole-code index and the amplitude reduction factor  $S_0^2$ . If the entry for  $S_0^2$  is less than 0.1, then  $S_0^2$  will be estimated from atomic overlap integrals. Experimental values of  $S_0^2$  are typically between 0.8 and 1.0. The defaults if the HOLE card is omitted are `ihole=1` for the  $K$  shell and  $S_0^2=1$ . The hole codes are shown in Table 2.2.

FEFF is designed to calculate absorption from completely filled shells. You can try to simulate absorption from valence electrons with FEFF, but you may get unreliable results. If you encounter difficulties and need valence shell absorption, please contact the authors.

For `ihole > 4`, the core-hole lifetime parameter ( $\gamma_{\text{ch}}$ ) is not tabulated in FEFF and is set equal to 0.1 eV, since the final state losses are then dominated by the self-energy. Use the EXCHANGE card to make adjustments ( $\gamma_{\text{ch}} = 0.1 + 2 \cdot \text{vi0}$ ).

```
* K-shell core hole, S02 estimated by overlap integrals
HOLE 1 0.0
```

#### ◆ POTENTIALS ipot Z [tag lmax1 lmax2 xnatph spinph]

*Standard*

The POTENTIALS card is followed by a list which assigns a unique potential index to each distinguishable atom. The potential index ipot is the index of the potential to be used for the phase shift calculation.

| index | edge | index | edge | index | edge | index | edge |
|-------|------|-------|------|-------|------|-------|------|
| 0     | N0   | 7     | M3   | 14    | N5   | 21    | O5   |
| 1     | K    | 8     | M4   | 15    | N6   | 22    | O6   |
| 2     | L1   | 9     | M5   | 16    | N7   | 23    | O7   |
| 3     | L2   | 10    | N1   | 17    | O1   | 24    | P1   |
| 4     | L3   | 11    | N2   | 18    | O2   | 25    | P2   |
| 5     | M1   | 12    | N3   | 19    | O3   | 26    | P3   |
| 6     | M2   | 13    | N4   | 20    | O4   |       |      |

Table 2.2: Available hole codes. The entries in the columns marked edge are written as they are recognized by the EDGE card. Index 0, N0, is the no hole option described in the NOHOLE card.

The required list entries are the unique potential index `ipot` and the atomic number `Z`. The `tag` is at most 6 characters and is used to identify the unique potential in the ‘`paths.dat`’ output file.

The optional list entries `lmax1` and `lmax2` are used to limit the angular momentum bases of the self-consistent potentials (XSPH) and full multiple scattering calculations (FMS). If a negative number (e.g., `lmax1 = -1`) is specified for either `lmax1` or `lmax2`, FEFF will automatically use a default based upon the atomic number of the species normally `lmax(atomic)`. The optional entry `xnatph` can be used to specify the stoichiometric number of each unique potential in the unit cell of a crystalline material. This helps in the calculation of the Fermi level. In the case of an infinite solid, `xnatph = 0.01` (default value) is a suitable value for the absorbing atom. The last optional entry `spinph` is used to specify the spin amplitude and relative spin alignment for spin-dependent calculations (see description of SPIN card for further usage details).

The absorbing atom must be given unique potential index 0. These unique potential indices are simply labels, so the order is not important, except that the absorbing atom is index 0, and you may not have missing indices (i.e., if you use index 3, you must also have defined unique potentials 1 and 2).

To save time the code calculates the overlapped atom potential for each unique potential only once, using as a sample geometry the first atom in the atom list with a given unique potential index. Thus it is essential that the neighborhood of that sample atom be representative. Failure to do so may cause the code to generate inaccurate potentials and phase shifts and poor XAS results.

It is often useful to assume that the potential for a given shell of atoms is the same as that of a previously calculated shell in order to save calculation time. For example, in Cu it is a good approximation to determine potentials only for the central atom and the first shell and to use the first shell potential (`ipot = 1`) for all higher shells. Such approximations should be checked in each case, however.

\* molecular SF6 Sulfur K edge, `lamx1=default`, `lmax2=3` (spdf basis)

## POTENTIALS

| * | ipot | Z  | tag | lmax1 | lmax2 | xnatph |
|---|------|----|-----|-------|-------|--------|
|   | 0    | 16 | S   | -1    | 3     | 1      |
|   | 1    | 9  | F   | -1    | 3     | 6      |

◆ **S02** s02*Standard*

The S02 card specifies the amplitude reduction factor  $S_0^2$ . If the entry for  $S_0^2$  is less than 0.1, then the value of  $S_0^2$  is estimated from atomic overlap integrals. Experimental values of  $S_0^2$  are typically between 0.8 and 1.0.

Alternatively, you can specify the value of  $S_0^2$  in the HOLE or EDGE card; however, the meaning of the parameters in the 'feff.inp' file is more clear if you use the S02 card.

```
* let FEFF calculate S02
S02    0.0
```

◆ **FOLP** ipot folp*Useful*

The FOLP card sets a parameter which determines by what factor the muffin-tin radii are overlapped. We recommend that the AFOLP card be used (default overlap = 1.15) in cases with severe anisotropy, and FOLP be used with caution, for example for Hydrogen or for fitting AXAFS. Typically only values larger than 1 and less than 1.3 should be used, except for hydrogen atoms, where we recommend the value 0.8. The AFOLP card is ignored when FOLP is used for a particular potential type.

```
* +20% overlap of muffin tin with unique potential 1
* -20% overlap of muffin tin with unique potential 2
FOLP 1  1.2    ! adjust overlap to fit AXAFS
FOLP 2  0.8    ! use 0.8 for hydrogen
```

◆ **EXCHANGE** ixc vr0 vi0 [ixc0]*Useful*

The EXCHANGE card specifies the energy dependent exchange correlation potential to be used for the fine structure and for the atomic background. *ixc* is an index specifying the potential model to use for the fine structure and the optional *ixc0* is the index of the model to use for the background function. The calculated potential can be corrected by adding a constant shift to the Fermi level given by *vr0* and to a pure imaginary “optical” potential (i.e., uniform decay) given by *vi0*. Typical errors in FEFF’s self-consistent Fermi level estimate are about 1 eV. (The CORRECTIONS card in Section 2.8 is similar but allows the user to make small changes in *vi0* and *vr0* after the rest of the calculation is completed, for example in a fitting process.) The Hedin–Lundqvist self-energy is used by default and appears to be the best choice for most applications we have tested in detail. The partially nonlocal model (*ixc*=5) gives slightly better results in some cases, but has not been tested extensively.

Another useful exchange model is the Dirac-Hara exchange correlation potential with a specified imaginary potential *vi0*. This may be useful to correct the typical error in non-self-

consistent estimates of the Fermi level of about +3 eV and to add final state and instrumental broadening.

Defaults if EXCHANGE card is omitted are: `ixc=0` (Hedin–Lundqvist), `vr0=0.0`, `vi0=0.0`. For XANES, the ground state potential (`ixc0=0`) is used for the background function and for EXAFS the Hedin–Lundqvist (`ixc0=0`) is used.

Indices for the available exchange models:

- 0 Hedin–Lundqvist + a constant imaginary part
- 1 Dirac–Hara + a constant imaginary part
- 2 ground state + a constant imaginary part
- 3 Dirac–Hara + HL imag part + a constant imaginary part
- 5 Partially nonlocal: Dirac–Fock for core + HL for valence electrons + a constant imaginary part

\*Hedin–Lundqvist -2eV edge shift and 1eV expt broadening  
EXCHANGE 0 2. 1.

\*Dirac–Hara exchange -3 eV edge shift and 5 eV optical potential  
EXCHANGE 1 3. 5.

#### ◆ NOHOLE

*Useful*

This card roughly simulates the effect of complete core-hole screening. It is useful to test the final state rule for calculated XAS, and to compare with other calculations (such as band structure or other codes) that do not have a core hole. The code will use as the final state that specified by the HOLE card for the matrix element calculation — the NOHOLE card will cause FEFF to calculate potentials and phase shifts as if there is no core hole. For *dDOS* and *L<sub>II</sub>* or *L<sub>III</sub>* absorption calculations, for example, NOHOLE often gives better agreement for white line intensities. Conversely NOHOLE tends to give poor XANES intensities for K-shell absorption in insulators.

#### ◆ RGRID delta

*Useful*

The radial grid used for the potential and phase shift calculation is

$$r(i) = \exp(-8.8 + (i - 1) \cdot \delta)$$

with `delta` = 0.05 by default. The default is sufficient for most cases. However, occasionally there are convergence problems in the atomic background at very high energies (the background curves upward) and in the phase shifts for very large atoms. If such convergence problems are encountered we suggest reducing `delta` to 0.03 or even 0.01. This will solve these problems at the cost of longer computation times (the time is proportional to  $1/\delta$ ). This option is also useful for testing and improving convergence of atomic background calculations.

RGRID 0.03 ! reduce grid for more accurate background at high energy

## ◆ SCF rfms1 [lfms1 nscmt ca nmix]

*Useful*

This card controls FEFF's automated self-consistent potential calculations. Thus all fields except rfms1 are optional. If this card is not specified then all calculations are done with the non self-consistent (overlapped atomic) potential. By default lfms1=0, nscmt=30 and ca=0.2.

## rfms1

This specifies the radius of cluster for full multiple scattering during the self-consistency loop. Typically one needs about 30 atoms within sphere specified by rfms1. Usually this value is smaller than the value rfms used in the FMS card, but should be larger than the radius of the second coordination shell.

## lfms1

The default value 0 is appropriate for solids; in this case the sphere defined by rfms1 is located on the atom for which the density of states is calculated. The value 1 is appropriate for molecular calculations and will probably save computation time, but may lead to inaccurate potentials for solids. When lfms1 = 1 the center of the sphere is located on absorbing atom.

## nscmt

This is the maximum number of iterations the potential will be recalculated. A value of 0 leads to non-self consistent potentials and Fermi energy estimates. A value of 1 also yields non-self consistent potentials but the Fermi energy is estimated more reliably from calculations of the LDOS. Otherwise, the value of nscmt sets an upper bound on the number of iterations in the self-consistency loop. Usually self-consistency is reached in about 10 iterations.

## ca

The convergence accelerator factor. This is needed only for the first iteration, since FEFF uses the Broyden algorithm to reach self-consistency. A typical value is 0.2; however, you may want to try smaller values if there are problems with convergence. After a new density is calculated from new Fermi level, the density after the first iteration is  $\rho_{\text{next}} = ca * \rho_{\text{new}} + (1 - ca) * \rho_{\text{old}}$ . ca = 1.0 is extremely unstable and should not be used.

## nmix

The nmix specifies how many iterations to do with mixing algorithm, before starting Broyden algorithm. The calculations of SCF in materials which contain f-elements may not converge. We encountered such problem for Pu. However, SCF procedure converged if we started Broyden algorithm after 10 iterations with mixing algorithm with ca=0.05.

\* Automated FMS SCF potentials for a molecule of radius 3.1 Angstroms

SCF 3.1 1

\* To reach SCF for f-elements and UNFREEZE we sometimes had to use

SCF 3.7 0 30 0.05 10

## ◆ UNFREEZEF

*Useful*

In many applications of *f*-electron systems, we found that it is usually preferable to freeze the *f*-electron density to that for atomic calculations in order to achieve well converged SCF potentials. This is the default in FEFF8.2. If one still wants to attempt calculating the *f*-DOS self-consistently, as in FEFF8.00 and 8.10, the UNFREEZEF card should be used.

```
* To include f-electrons in SCF calculations use
  UNFREEZEF
```

## ◆ INTERSTITIAL inters vtot

*Advanced*

The construction of interstitial potential and density may be changed by using INTERSTITIAL card. `inters = ipot + 2*irav + 6*irmt`. `ipot=1` might be useful when only the surroundings of the absorbing atom are specified in 'feff.inp'. `irav` and `irmt` are described only for completeness and nonzero values are strongly not recommended.

`ipot`

defines how to find interstitial potential: `ipot=0` (default) the interstitial potential is found by averaging over the entire extended cluster in 'feff.inp'. `ipot = 1` the interstitial potential is found locally around absorbing atom.

`irav`

also changes how interstitial potential is found. 0 (default) equation for  $V_{int}$  is constructed at  $r_{av}=r_{nrm}$ , 1 - at  $r_{av}=(r_{mt}+r_{nrm})/2$ , 2 - at  $r_{av}=r_{mt}$

`irmt`

0 : Norman prescription for *mt* radii (default) 1 : Matching point prescription for *mt* radii (do not use)

`vtot`

is the volume per atom normalized by  $ratmin^3$  ( $vtot=(\text{volume per atom})/ratmin^3$ ), where *ratmin* is the shortest bond for the absorbing atom. This quantity defines total volume (needed to calculate interstitial density) of the extended cluster specified in 'feff.inp'. If  $vtot \leq 0$  then the total volume is calculated as a sum of norman sphere volumes. Otherwise,  $totalvolume = nat * (vtot * ratmin^3)$ ; where *nat* is a number of atoms in extended cluster. Thus  $vtot=1.0$  is appropriate for cubic structures, such as NaCl. The INTERSTITIAL card may be useful for open systems (e.g. those which have ZnS structure).

```
* improve interstitial density for ZnS structures.
* vtot = (unit_cell_volume/number_of_atoms_in_unit_cell)/ratmin**3)=1.54
INTERSTITIAL 0 1.54
```

## ◆ ION ipot ionization

*Advanced*

The ION card ionizes all atoms with unique potential index *ipot*. Negative values and non-

integers are permitted, however ionicities larger than 2 and less than  $-1$  often yield unphysical results. Our experience with charge transfers using the SCF card suggests values for `ionization` about 5–10 times smaller than the formal oxidation state. The ION card is probably not needed if the potential is self-consistent. However, it can be used to put some total charge on a cluster. In this case we suggest using the same ionicity for all atoms in cluster (i.e. total ionization divided by number of atoms). For example, for diatomics like Br<sub>2</sub>, the fully relaxed configuration has a formal ionization of 1 on the scattering atom. Because of charge transfer, the actual degree of ionization is will be much smaller. In non-self-consistent calculations the default (non-ionized) scattering potentials are often superior to those empirically ionized, and the results should be checked both ways. The default if ION cards are omitted is that the atoms are not ionized.

```
* Simulates effective ionization for formal valence state +1
* ipot, ionization
  ION  1  0.2
```

#### ◆ SPIN `ispin [x y z]`

*Advanced*

The SPIN card accommodates spin-dependent calculations. This card is used to specify the type of spin-dependent calculation (`ispin`) for spin along (x, y, z) direction. By default the spin is assumed along z-axis. The SPIN card is required for the calculation of all spin-dependent effects, including XMCD and SPXAS (see Section 4.4.3).

Default `ispin = 0` is used for spin independent calculations. Values `ispin = 1` and `ispin = -1` are used for XMCD calculations, while `ispin = 2` and `ispin = -2` are used to calculate spin-polarized LDOS and SPXAS. In all cases, the positive values correspond to the spin-up portion of the calculations, and the negative values to the spin-down portion.

Automatic spin-dependent calculations require that the FEFF code be compiled with `nspin = 2` (found in `'/src/HEADERS/dim.h'`). This will also add the contribution from spin-flip processes (which we find typically very small), but will require 4 times the memory and 8 times the execution time for XANES region. Thus in general we would recommend the use of `nspin = 1` dimension, unless spin-flip processes are expected to be significant (e.g. for 5f elements).

Whenever spin-dependent calculations are used, the spin amplitude and relative spin orientation should be specified in `'feff.inp'`, in the 7th column of the POTENTIALS card.

The spin-dependent potentials are calculated from the spin-dependent densities, using von Barth-Hedin results for the uniform electron gas. We use the rough prescription to construct the spin-dependent muffin-tin potential. This should be fine for EXAFS where small details of the potential are irrelevant, but may not be good enough in the XANES region, where the self-consistent spin-dependent muffin-tin potential can lead to better results.

```
* Make spin-up XMCD calculations (XMCD card must also be included)
SPIN 1
XMCD
```

POTENTIALS card for spin-dependent calculations:

```

* GdFeO, spin-dependent
* Note that SPIN card must also be present in feff.inp, as well as
* additional cards specific to desired calculations (in this case, XMCD)
POTENTIALS
*  ipot  z  tag  lmax1  lmax2  xnatph  spamp
    0   64  Gd   3      3      0.1    7.0    (c sublattice)    up
    1   26  Fe   2      3      2      4.0    (a sublattice)    up
    2   26  Fe   2      3      3     -4.0    (d sublattice)    down
    3    8   O   2      3     12.0    0.0    (h sublattice)
    4   64  Gd   3      3      2.9    7.0    (c sublattice)    up

```

## 2.4 XSPH: Cross-section and phase shifts

Relativistic dipole matrix elements (alpha form) are calculated using atomic core and normalized continuum wave functions. Polarization dependence is optionally incorporated in the dipole-operator. Scattering phase shifts are determined by matching at the muffin-tin radius. Additionally,  $\ell$ -projected density of states can be calculated in this module, but it is of limited quality due to finite cluster calculations and neglect of nonspherical corrections.

### ◆ ELLIPTICITY *ellipticity x y z*

*Useful*

This card is used with the POLARIZATION card (see below). The `ellipticity` is the ratio of amplitudes of electric field in the two orthogonal directions of elliptically polarized light. Only the absolute value of the ratio is important for nonmagnetic materials. The present code can distinguish left- and right-circular polarization only with the XMCD or XNCD cards. A zero value of the ellipticity corresponds to linear polarization, and unity to circular polarization. The default value is zero.

`x, y, z` are coordinates of any nonzero vector in the direction of incident beam. This vector should be approximately normal to the polarization vector.

```
* Average over linear polarization in the xy-plane
```

```
ELLIPTICITY 1.0 0.0 0.0 -2.0
```

### ◆ POLARIZATION *x y z*

*Useful*

This card specifies the direction of the electric field of the incident beam or the main axis of the ellipse in the case of elliptical polarization. `x, y, z` are the coordinates of the nonzero polarization vector. The ELLIPTICITY card is not needed for linear polarization. If the POLARIZATION card is omitted, spherically averaged xafs will be calculated.

Note that polarization reduces the degeneracy of the paths, increasing the calculation time. Choosing polarization in the directions of symmetry axes will result in a faster calculation.

```
POLARIZATION 1.0 2.5 0.0
```

◆ **MULTIPOLE** *le2 [l2lp]**Useful*

Specifies which multipole transitions to include into the calculations. Only dipole: *le2*=0 (default), dipole and quadrupole (*le2*=2), dipole and magnetic dipole (*le2*=1).

Additional field *l2lp* can be used to calculate individual dipolar contributions coming from  $L \rightarrow L + 1$  (*l2lp*=1) and from  $L \rightarrow L - 1$  (*l2lp*=-1). Notice that in polarization dependent data there will be also a cross term, which will be calculated only when *l2lp*=0.

```
MULTIPOLE  2  0  *combine dipole and quadrupole transitions.
MULTIPOLE  0 -1  *calculate dipolar L -> L-1 transitions
```

◆ **LDOS** *emin emax eimag**Useful*

The angular momentum projected density of states is placed by default on a standard grid currently fixed at 84 points. *emin* and *emax* are the minimum and maximum energies of the  $\ell$ DOS calculation and *eimag* is the imaginary part of potential used in the calculations. This is equivalent to Lorentzian broadening of the  $\ell$ DOS with half-width = *eimag*. If *eimag* is negative, the code automatically sets it to be 1/3 of the energy step. The output will be written again into 'ldosNN.dat' files. To obtain LDOS you must run the second module by setting the second CONTROL argument to 1. If 84 points are not enough, you can divide the energy range by 2 and run the code twice. The LDOS card is very useful when examining densities of states for interpreting XANES or when the self-consistency loop fails or gives very strange results. For crystals our LDOS will always be broadened due to the effect of finite cluster size.

```
*      emin emax eimag
LDOS  -20  20  0.2
```

◆ **EXAFS** [*xkmax*]*Standard*

The EXAFS card is used to change the maximum value of  $k$  for EXAFS calculations. Default value is  $20 \text{ \AA}^{-1}$ . Now code can calculate even to higher values, however user will be prompted to increase dimensions in 'dim.h' file and recompile the code. For high  $k$  calculations it might be necessary to make smaller steps using RGRID card.

```
EXAFS 25
```

◆ **XANES** [*xkmax xkstep estep*]*Standard*

The XANES card is used when a calculation of the near edge structure including the atomic background and absolute energies are desired.

The XANES calculation is currently limited to the (extended) continuum spectrum beyond the Fermi level. Thus bound states are not generally included; however, in molecules weakly bound states that are below the vacuum but above the muffin-tin zero will show up as resonances. The absolute energies are based on atomic total energy calculations using the

Dirac-Fock-Desclaux atom code. The accuracy of this approximation varies from a few eV at low  $Z$  to a few hundred eV for very large  $Z$ . All parameters are optional. Default: XANES not calculated unless card is present.

The optional parameters are used to change the output energy mesh for the XANES calculation. `xkstep` specifies the size of the output  $k$  grid far from the edge. `xkmax` is the maximum  $k$  value of the XANES calculation. FMS calculations are not accurate beyond about  $k = 6$ ; for larger values of  $k$ , e.g.  $k = 20$  with the path expansion, FMS must be turned off. The grid at the edge will be regular in energy with a step size of `estep`. The default values are `xkstep` = 0.07, `xkmax` = 8, and `estep` =  $\gamma_{\text{ch}}/4 + \text{vi0}/2$ , where `vi0` is given by the EXCHANGE card described in Section 2.3.

```
* finer grid for XANES calculation
XANES 6. .05 .3
```

#### ◆ TDLDA ixfc

*Useful*

Uses time-dependent local density approximation (TDLDA) theory to account for screening of the x-ray field and of the photoelectron–core-hole interaction. The parameter `ixfc` determines whether static or dynamic screening is used. `TDLDA 0` (static screening) accounts for screening of the x-ray field (but not the field of the core-hole), blue-shifting the spectrum. `TDLDA 0` seems to work well at high energies. `TDLDA 1` (dynamic screening) accounts for screening of the x-ray and core-hole fields, blue-shifting the spectrum less than `TDLDA 0`. The TDLDA card affects only module 2.

TDLDA theory takes into account polarization-type many body effects (i.e., polarization of the electronic charge) which screen the local x-ray field. These effects are most important for xrays with energies less than 1 keV, hence `TDLDA 0` works well at high energies. The screened interaction is calculated partially based on the Bethe-Salpeter equation, in the basis of local atomic states. This approximation yields efficient calculations of the spectra in terms of screened transition matrix elements. Note that TDLDA does not account for core-hole relaxation effects.

L-shell x-ray absorption in 3d transition metals is sensitive to dynamic screening effects. For rare-gas solids, dynamic screening accounts for deviations of the  $L_3/L_2$  intensity branching ratio from the 2:1 value of independent-electron theory.

See A.L. Ankudinov, A.I. Nesvizhskii, and J.J. Rehr, *Dynamic screening effects in x-ray absorption spectra*, Phys. Rev. B 67, 115120 (2003) for further details on the implementation of TDLDA in the FEFF calculations. See also A.I. Nesvizhskii's thesis, Ch. 8 (pp. 82-105) (available on the FEFF documents webpage) for a brief description of the TDLDA theory, or the original paper by A. Zangwill and P. Soven, Phys. Rev. A, 21:1561, 1980 (note that the original paper handles TDLDA differently from the approach used in FEFF).

```
* use dynamic screening in xsph, this will only impact module 2 calculations
TDLDA 1
```

◆ **XMCD or XNCD** [xkmax xkstep estep]*Advanced*

Use either of the cards to calculate x-ray circular dichroism (the output will contain both magnetic and natural). The code calculates XMCD and XNCD from specified edge and grid, specified by auxiliary fields exactly as in XANES card.

For nonmagnetic systems only XNCD will be present, while for magnetic materials with high symmetry only XMCD is present. Both will be present for magnetic materials with low symmetry, and x-ray direction (ELLIPTICITY card) must be used to disentangle the two contributions.

The XNCD originates from cross dipole-quadrupole contributions for certain nonmagnetic materials, such as special types of crystals. It will change sign for opposite direction of propagation (use the ELLIPTICITY card to do this). We performed calculations for  $\text{LiIO}_3$  and found results very similar to previous multiple scattering XNCD calculations. XNCD requires that the XANES card also be used.

The XMCD (dipolar and quadrupolar) does not change sign under the change of direction of x-ray propagation, and is zero for nonmagnetic systems. The origin of the effect is that due to spin-orbit coupling the right circular polarized light will produce more electrons with spin along or opposite to the direction of x-ray propagation. Thus it is important to use spin-dependent calculations for XMCD calculations. See the SPIN card and Section 4.4.2 for more details on spin-dependent calculations and an example for XMCD.

Note that the XMCD signal will only be contained in the output if the FEFFcode has been compiled with `nspx = 2`. For the (default) value `nspx = 1`, you have to combine data from two ‘xmu.dat’ files. A simple program to do this, ‘spin.f’ is available on the FEFF web site, and printed in Section 4.4, where you can also find more details on the signal extraction process.

The EXAFS region can be used to determine the position of spins relative to magnetic field. The XMCD card has to be present in ‘feff.inp’ for these calculations with FEFF8.40.

For antiferromagnets, the XMCD should be zero, and a measure of the spin-up and spin-down signals can be accomplished using SPXAS. See Section 4.4.3 for more details and an example.

◆ **DANES** [xkmax xkstep estep]*Advanced*

To calculate x-ray scattering amplitude  $f'$  instead of absorption  $f''$ , including solid state effects. Calculate contribution from specified edge and grid, which is specified as in the XANES card. This card is still experimental.

◆ **FPRIME** emin emax estep*Advanced*

To calculate x-ray scattering factor  $f'$  far from the edge (only atomic part). The energy grid is regular in energy with estep between emin and emax. This is typically needed to find out contribution from other edges to the edge calculated with DANES card. Later it may be automated. Total scattering amplitude  $f'(Q, E) = f_0(Q) + f'(E) + if''(E)$ . In the dipole approximation  $f'$  and  $f''$  do not depend on  $Q$ , but this does not hold with quadrupole transitions added. This is currently neglected and  $f'(E) = \text{DANES}(\text{edge}) + \text{FPRIME}(\text{all other edges}) + \text{total energy term}$  in ‘fpf0.dat’.  $f_0(Q)$  is also tabulated in ‘fpf0.dat’;  $f''$  is printed out by

FPRIME in electronic units, and can be used to obtain total  $f'$ . The total energy correction to  $f'$  is given in first line of 'fpr0.dat' in Cromer-Liberman, and the more accurate Kissel-Pratt form. See our paper on elastic scattering amplitude for references and details.

◆ **XES** `emin emax estep`

*Advanced*

To calculate nonresonant x-ray emission spectra (XES) for a specified grid. XES may be compared to the occupied DOS.

◆ **RPHASES**

*Advanced*

If this card is present, real phase shifts rather than complex phase shifts will be used. The results of the calculation will not be accurate. This option is intended to allow users to obtain real scattering phase shifts for use with other programs, or for diagnostic purposes. The phase shifts can be written to output files 'phaseNN.dat' using the PRINT card. If the RPHASES card is present, these will be the real phase shifts.

◆ **RSIGMA**

*Advanced*

If this card is present, the imaginary part of self-energy will be neglected. It might be useful for calculations in XANES region, since the imaginary part of Hedin -Lundqvist self-energy tends to overestimate losses in this region.

## 2.5 FMS: Full multiple scattering

This module carries out a full multiple scattering XANES calculation for a cluster centered on the absorbing atom. Thus all multiple-scattering paths within this cluster are summed to infinite order. This is useful for XANES calculations, but usually cannot be used for EXAFS analysis. FMS loses accuracy beyond  $k = (l_{\max} + 1)/r_{\text{mt}}$ , which is typically about  $4 \text{ \AA}^{-1}$  since the muffin-tin radius  $r_{\text{mt}}$  is typically about  $1 \text{ \AA}$ .

◆ **FMS** `rfms lfms2 [minv toler1 toler2 rdirec]`

*Standard*

Compute full multiple scattering within a sphere of radius `rfms` centered on the absorbing atom. If you don't use FMS card, the multiple scattering path expansion is used.

`rfms` is the cluster radius used in all modules but POT. Specifically is used for in the LDOS, FMS, and as the lower limit of pathfinder calculations. Typically a converged XANES calculation requires about 50-150 atoms in a cluster. The FMS module sums all MS paths within the specified cluster. The number of atoms in this cluster is limited to 87 by default, but one can manually increase the dimension parameter `nclusx` in the ancillary FEFF8 source code file 'dim.h' to alter the maximum cluster size. If there are more than `nclusx` atoms within the specified cluster size, FEFF will reduce the cluster size and issue a warning.

For EXAFS analysis one typically calculates to  $k = 20$ , but FMS results are not accurate at high energies. Thus if you are running FEFF8 for EXAFS you should not use FMS and XANES cards. It is, however, desirable to calculate self-consistent potentials even for EXAFS

calculations as in the example below:

```
*calculate EXAFS with SCF potentials and paths to R=6 angstroms
CONTROL 1 1 1 1 1 1
*fms
SCF 3.1
RPATH 6.0
EXAFS
```

If the value of RPATH as described in Section 2.6 is greater than `rfms`, the pathfinder will look for paths which extend beyond the cluster used for the FMS and add them to the FMS calculation of the  $\ell$ DOS and XANES:

$$G_{\text{tot}} = G_{\text{fms}} + G_0 t_i G_0 + G_0 t_i G_0 t_j G_0 + \dots$$

where at least one atoms  $i$  in the path is outside the FMS cluster and the value of RPATH is the maximum half path length for LDOS, FMS and pathfinder modules. Note: this approximation may not be accurate and should be used with caution.

The MS expansion sometimes does not converge well in the XANES energy region. Thus one should avoid adding paths for LDOS and XANES, and RPATH should be less than `rfms`. Adding single scattering path only (NLEG 2) usually works well to check the convergence of FMS. But adding double scattering (NLEG 3) often leads to very bad results in XANES. Thus RPATH is useful for EXAFS or for XANES only when the path expansion is stable.

The optional `lfms2` argument is a logical flag which defines how the FMS is done, similar to the flag `lfms1` in the SCF card. With the default value of 0 (appropriate for solids), the FMS is calculated for a cluster of size `rfms` around each representative unique potential. With `lfms=1` (appropriate for molecules), FMS is done only once for a cluster of size `rfms` around absorbing atom only. The proper use of this flag can lead to a considerable time savings.

For example, if you calculate FMS for a molecule smaller than 40 atoms, there is no need to invert `nph + 1` matrices, and `lfms1 = 1` will reduce time for calculations by factor (`nph + 1`), where `nph` is a number of unique potentials listed in POTENTIAL card).

A typical use of the FMS card uses `lfms2 = 0`, for example,

```
FMS 6.0      ! for XANES and LDOS need about 100 atom cluster
RPATH 8.0    ! usually use rpath < rfms
NLEG 2       ! adds 2 leg paths between 6 and 8 angstroms
```

For molecules of less than 30 atoms of radius 4.0 Å we suggest using `lfms2 = lfms1 = 1`, as in

```
FMS 5.0 1
RPATH -1
```

The optional `minv` index defines the FMS algorithm used in the calculations. By default (`minv=0`) the FMS matrix inversion is performed using LU decomposition. However, several alternative have been designed for FMS algorithm that start to work faster than LU decomposition for clusters of more than a 100 atoms. (See the `FEFF8.2` reference). We strongly recommend the Lanczos recursion method (`minv=2`) which is very robust and speeds the calculations by a factor of 3 or more. The Broyden algorithm (`minv=3`) is faster, but less reliable, and may fail to converge, if the FMS matrix has large eigenvalues.

The optional `toler1` defines the tolerance to stop recursion and Broyden algorithm. The default value 0.001 gives results in agreement with LU decomposition to within a linewidth.

The optional `toler2` sets the matrix element of  $Gt$  matrix to zero if its value is less than `toler2` (default 0.001).

The optional `rdirec` sets the matrix element of the  $Gt$  matrix to zero if the distance between atoms is larger than `rdirec`.

The last two variables can make the matrix  $Gt$  very sparse so both recursion and Broyden algorithms work faster. For example for large Si calculations with the Lanczos algorithm, we used

```
FMS 29.4 0 2 0.001 0.001 40.0
```

#### ◆ DEBYE temp thetad [idwopt]

*Useful*

See the full description in Section 2.8 for details. The effect of temperature on FMS is approximated by multiplying each free propagator by  $\exp(-\sigma^2 k^2)$ , which gives correct DW factors for single scattering. The DW factors for multiple scattering are not exact, but their contribution is reduced both by thermal factors and by the mean free path. Also if you are running the FMS module, then you can only obtain XANES, where this approximate treatment of thermal effect is probably adequate.

## 2.6 PATHS: Path enumeration

The code uses a constructive algorithm with several path importance filters to explore all significant multiple-scattering paths in order of increasing path length. The paths are determined from the list of atomic coordinates in `'feff.inp'`. An efficient degeneracy checker is used to identify equivalent paths (based on similar geometry, path reversal symmetry, and space inversion symmetry). To avoid roundoff errors, the degeneracy checker is conservative and occasionally treats two degenerate paths as not degenerate. These errors occur in the third or fourth decimal place (less than 0.001 Ang) but are fail safe; that is, no paths will be lost. Of course, all paths which are completely inside the FMS cluster are automatically excluded from paths list.

The criteria used in filtering are based on increasingly accurate estimates of each path's amplitude. The earliest filters, the pathfinder heap and keep filters, are applied as the paths

are being searched for. A plane wave filter based on the plane wave approximation (plus a curved wave correction for multiple-scattering paths) and accurate to about 30% is applied after the paths have been enumerated and sorted. Finally, an accurate curved wave filter is applied to all remaining paths.

#### ◆ PCRITERIA keep-criterion heap-criterion

*Advanced*

These criteria, like those described in the CRITERIA card, also limit the number of paths. However, they are applied in the pathfinder and eliminate unimportant paths while the pathfinder is doing its search. The pathfinder criteria (pcrit's) do not know the degeneracy of a path and are therefore much less reliable than the curved wave and plane wave criteria in the CRITERIA card above. These path finder criteria (keep and heap) are turned off by default, and we recommend that they be used only with very large runs, and then with caution.

The keep-criterion looks at the amplitude of chi (in the plane wave approx) for the current path and compares it to a single scattering path of the same effective length. To set this value, consider the maximum degeneracy you expect and divide your plane wave criterion by this number. For example, in fcc Cu, typical degeneracies are 196 for paths with large  $r$ , and the minimum degeneracy is 6. So a keep criterion of 0.08% is appropriate for a pw criteria of 2.5%.

The heap-criterion filters paths as the pathfinder puts all paths into a heap (a partially ordered data structure), then removes them in order of increasing total path length. Each path that is removed from the heap is modified and then considered again as part of the search algorithm. The heap filter is used to decide if a path has enough amplitude in it to be worth further consideration. If we can eliminate a path at this point, entire trees of derivative paths can be neglected, leading to enormous time savings. This test does not come into play until paths with at least 4 legs are being considered, so single scattering and triangular (2 and 3 legged) paths will always pass this test. Because only a small part of a path is used for this criterion, it is difficult to predict what appropriate values will be. To use this (it is only necessary if your heap is filling up, and if limiting rpath doesn't help), study the results in 'crit.dat' from runs with shorter rpath and experiment with the heap criterion accordingly. In the future, we hope to improve this filter.

Before using these criteria, study the output in the file 'crit.dat' (use print option 1 for paths, see Table 2.1), which has the values of critpw, keep factor and heap factor for all paths which pass the critpw filter.

Default: If this card is omitted, the keep and heap criteria are set to zero, that is, no filtering will be done at this step in the calculation.

```
* fcc Cu had degeneracies from 6 to 196, so correct for this by
* dividing pw-crit of 2.5% by 30 to get 0.08 for keep crit. Check this
* empirically by running with pcrits turned off and studying crit.dat.
* After studying crit.dat, choose 0.5 for heap crit.
PCRITERIA 0.08 0.5
```

#### ◆ RPATH rpath

*Useful*

The RPATH card determines the maximum effective (half-path) distance, `rpath`, of a given path. RPATH is equivalent to the RMAX card in the FEFF7 code. We changed the name because it provides a clearer distinction between the max distance in the MS path expansion and that for FMS calculations. Typically `rpath` is needed for EXAFS calculations only to set limits on the number of calculated paths. Note that `rpath` is one-half of the total path length in multiple-scattering paths. Setting this too large can cause the heap in the pathfinder to fill up. Default is `rpath` = 2.2 times the nearest neighbor distance. Since the multiple scattering expansion is unstable close to the absorption edge, the path (MS) expansion should be used only for EXAFS calculations or for diagnosing the XANES or LDOS calculations. If you use FMS for XANES calculations, better results are obtained without the MS contribution. For EXAFS analysis this card is extremely useful, since `rpath` cuts off long paths which give contribution only at high R values in R-space.

```
* include MS paths with effective length up to 5.10 Ang
RPATH      5.10
```

#### ◆ SS index ipot deg rss

*Advanced*

The SS card can *only* be used with OVERLAP cards when the atomic structure is unknown but one does know the distance and coordination numbers and wants to generate an approximate EXAFS contribution. Thus the pathfinder cannot be used in this case. Instead the user has to specify explicitly the single scattering paths and their degeneracy. OVERLAP cards must be used to construct the potentials for the use with SS card. The parameters are a shell index, which is a label used for 'feffNNNN.dat' file name, a unique potential index `ipot` identifying the unique potential of the scattering atom, the degeneracy (or multiplicity) of the single scattering path, and the distance to central atom `rss`.

This information is used to write the file 'paths.dat' and is not needed when ATOMS card is used. To generate SS paths with ATOMS use (NLEG 2) card.

```
* index ipot deg rss generate single scattering results
SS 29 1 48 5.98 parameters for 19th shell of Cu
```

## 2.7 GENFMT: XAFS parameters

For each path the code calculates the effective scattering amplitude ( $f_{\text{eff}}$  from which FEFF gets its name, see Section 3.2.4) and the total scattering phase shift along with other XAFS parameters using the scattering matrix algorithm of Rehr and Albers. Once the scattering phase shifts and the paths are determined, no other input is necessary for this calculation.

#### ◆ CRITERIA critcw critpw

*Useful*

Since the number of multiple scattering paths gets large very quickly, it is necessary to eliminate as many paths as possible. Fortunately, we have found that most multiple scattering

paths have small amplitudes and can be neglected. Various cutoff criteria are used in FEFF8 to limit the number of paths to consider. These criteria are based on the importance of the path, defined as the integral over the full energy range of  $\chi(k) \cdot dk$ . Very close to the edge these cutoff criteria should be examined with care and in some cases reduced from the values used for EXAFS.

**critcw** is the cutoff for a full curved wave calculation. A typical curved wave calculation requires a complete spherical wave calculation, which typically takes seconds of CPU time per path. The default value of **critcw** is 4%, meaning that any path with mean amplitude exceeding 4% of largest path will be used in calculation of  $\chi$ . The criterion **critcw** is used by GENFMT. Since the XAFS parameter calculation is already done, the savings is not in computer time, but in disk space and ease of analysis. The values of **critcw** for each path are written in the file 'list.dat' written by module GENFMT.

**critpw** is a plane-wave approximation to  $\chi$ . This is extremely fast to calculate, and is used in the pathfinder. The default value of **critpw** is 2.5, meaning that any path with mean amplitude 2.5% of largest path, including degeneracy factors, (in plane wave approximation) will be kept. Any path that does not meet this criterion will not be written to 'paths.dat', and there is no need to calculate the XAFS parameters for this path. The default for **critpw** is less than that for **critcw** since some paths are more important when the full curved wave calculation is done than they appear in the plane wave approximation. Since the plane wave estimate is extremely fast, use this to filter out as many paths as you can. The file 'crit.dat' (written by the module PATHS) tells you **critpw** for each path that passes the criterion.

The method of calculation of these importance factors has been improved for FEFF8, so don't worry if the values for some paths have changed slightly from previous versions. (Default values **critcw**=4.% **critpw**=2.5%)

```
CRITERIA  6.0  3.0  * critcw 6%, critpw 3%
CRITERIA   0   0   * use all paths (cw and pw criteria turned off)
```

#### ◆ NLEG nleg

*Useful*

The NLEG card limits the number of scattering paths to **nleg**. If **nleg** is set to 2, only single scattering paths are found. The default is **nleg** = 8.

```
* only single scattering paths (i.e. 2 legged paths)
NLEG 2
```

#### ◆ IORDER iord

*Advanced*

Order of the approximation used in module GENFMT. FEFF uses order 2 by default which is correct to terms of order  $1/(pR)^2$ , and corresponds to 6x6 scattering matrices in the Rehr-Albers formalism. Single scattering is calculated exactly to this order. The 6x6 approximation is accurate to within a few percent in every case we have tried (that is, higher order doesn't change the result more than a few percent). However  $M_{IV}$  shells and higher shells may require

increased `iorder` for coupling the matrix elements. Changing the default values requires some familiarity with the Rehr–Albers paper and the structure of the module GENFMT. To do so, follow the instructions in the FEFF source code in subroutine `setlam`. The key `iord` is passed to `setlam` for processing. You may need to change the code parameter `lamtot` if you want to do higher order calculations. For details of the algorithm used by GENFMT, see the paper by J.J. Rehr and R.C. Albers (see the references in Appendix C). For the  $M_{IV}$  and higher edges, you may receive the error message like: **Lambda array overfilled**. In that case the calculations should be repeated with `IORDER -70202` (10x10 matrices).

```
* change iorder for M4 calculations
IORDER -70202
```

#### ◆ NSTAR

*Advanced*

When this card is present, GENFMT will write the file ‘`nstar.dat`’ with the effective coordination number  $N^*$  which is the coordination number weighted by  $\cos^2(\theta)$  to correct for polarization dependence in SEXAFS calculations.

## 2.8 FF2CHI: XAFS spectrum

The module FF2CHI constructs the XAS spectrum  $\chi(k)$  or  $\mu$  using the XAFS parameters described in Section 3.2.4 from one or more paths, including any FMS contributions. Single and multiple scattering Debye–Waller factors are calculated using, for example, the correlated Debye model. Output from this module is the total XAFS spectrum and optionally, the contribution to the XAFS from each path individually. Numerous options for filtering, Debye–Waller factors, and other corrections are available.

#### ◆ DEBYE temperature Debye-temperature [idwopt]

*Standard*

The Debye card is used to calculate Debye–Waller factors for each path using the correlated Debye Model. The model is best suited for homogeneous systems, where it is quite accurate. CAUTION: in heterogeneous systems the model only gives approximate values which can easily be off by factors of two or more. Temperatures are in kelvin. If this card is present, the correlated Debye model Debye–Waller factors will be summed with the DW factors from the SIG2 card and from the ‘`list.dat`’ file, if present.

```
*Debye-Waller factors for Cu at 190K with correlated Debye Model
DEBYE 190 315
```

By default, `idwopt=0` specifies that the correlated Debye model is used to calculate EXAFS Debye–Waller factors. Two additional models for calculating DW factors are available in FEFF8 based on the information about the harmonic force constants in the material. `idwopt=1` means the equation of motion (EM) method is used to get Debye–Waller factors and `idwopt=2` means the recursion method (RM) which is an improved correlated Einstein model. Both methods

are faster than molecular dynamics simulations, and the recursion method is much faster than the equation of motion method. However, the equation of motion method leads to somewhat more accurate results than the recursion. These additional methods seem to be superior to correlated Debye model in cases with tetrahedral coordination, such as solid Ge or many biological materials. Both EM and RM methods need additional input (the force constants) and a complete description of both is given in Anna Poiarkova's thesis (see the FEFF Project web site (<http://feff.phys.washington.edu>) and in the associated documentation.

```
* Calculate Debye-Waller factors for Cu at 190K with equation of motion
DEBYE 190 0 1
```

◆ **CORRECTIONS** real-energy-shift imaginary-energy-shift *Useful*

The real energy shift moves  $E_0$  in the final  $\chi(k)$  and the imaginary energy shift adds broadening to the result. The real energy shift is useful to correct the error in FEFF's Fermi level estimate and the imaginary part can be used to correct for experimental resolution or errors in the core-hole lifetime. This error in the Fermi level is typically about 1 eV with self-consistent calculations and about 3 eV with overlapped atom potentials. The imaginary energy is typically used to correct for instrument broadening or as a correction to the mean free path calculated by FEFF. This affects only the module FF2CHI, which combines the results in all of the 'feffNNNN.dat' files. This card is useful in fitting loops because you can simply make such energy corrections and see the results without redoing the entire XAFS parameter calculation. CAUTION: the results are not as accurate as those obtained with the EXCHANGE card. Both energies are in eV. (See also the EXCHANGE card in Section 2.3).

```
* Reduce E0 by 3.0 eV and add 1 eV of broadening (full width)
* This will only affect module 4, ff2chi
CORRECTIONS 3.0 1.0 real shift, imag shift
```

◆ **SIG2** sig2 *Useful*

Specify a global Debye-Waller factor to be used or added to Debye-Waller calculations (see DEBYE card) for all paths. This value will be summed with the correlated Debye model value (if the DEBYE card is present) and any value added to 'list.dat'. Units are  $\text{\AA}^2$ . This card can be used, for example to add Debye-Waller factors from structural disorder.

```
SIG2 0.001 add 0.001 globally to all DW factors
```

## Chapter 3

# Input and Output Files

Other files required by the various modules are created by FEFF from ‘feff.inp’. Some of these other files may be edited by the user as a way to modify the input data to the modules, see Section 3.3. See the PRINT card in Section 2.2 to obtain various diagnostic files. Section 3.1 summarizes this structure, the rest of this section describes the structure in more detail.

### 3.1 Module Input and Output Files

**Module 0** ..... RDINP

**Purpose of Module:** Read input data

**Input files:** ‘feff.inp’

**Output files:** ‘geom.dat global.dat modN.inp (N=1-6)’

**Other output:** ‘paths.dat’ (only if SS card is used)

**Description:** Reads the ‘feff.inp’ file, makes appropriate operations on the data, and writes resulting information into several output files, that contain formatted data needed for all modules.

**Module 1** ..... POT

**Purpose of Module:** Calculate embedded atomic potentials for the photoelectron

**Input files:** ‘mod1.inp’ and ‘geom.dat’

**Output files:** ‘pot.bin’

**Other output:** diagnostic files (see Table 2.1 on page 9)

**Description:** Reads ‘mod1.inp’ and calculates potentials for the photoelectron which are written into ‘pot.bin’. Optionally, POT will write other diagnostic files with information about the potentials.

**Module 2** ..... XSPH

**Purpose of Module:** Calculate cross-section and phase shifts

**Input files:** 'mod2.inp', 'geom.dat', 'global.dat' and 'pot.bin'

**Output files:** 'phase.bin', and 'xsect.bin',

**Other output:** diagnostic files (see Table 2.1 on page 9), 'axafs.dat', and 'ldosNN.dat' ( $\ell$ DOS)

**Description:** XSPH writes the binary file 'phase.bin', which contains the scattering phase shifts and other information needed by PATHS and GENFMT. The atomic cross-section data is written in 'xsect.bin' and used in final module (FF2CHI) for overall normalization. Optionally, XSPH will write other diagnostic files with information about the phase shift calculations.

### Module 3 ..... FMS

**Purpose of Module:** Calculate full multiple scattering for XANES and  $\ell$ DOS

**Input files:** 'mod3.inp', 'global.dat', 'geom.dat', and 'phase.bin',

**Output files:** 'fms.bin'

**Other output:**

**Description:** Performs full multiple scattering algorithm. Writes output into 'fms.bin' for the FF2CHI module, which contains the  $\chi(k)$  from FMS.

### Module 4 ..... PATHS

**Purpose of Module:** Path enumeration

**Input files:** 'mod4.inp', 'geom.dat', 'global.dat' and 'phase.bin'

**Output files:** 'paths.dat'

**Other output:** 'crit.dat'

**Description:** PATHS writes 'paths.dat' for use by GENFMT and as a complete description of each path for use of the user. PATHS will optionally write other diagnostic files. The file 'crit.dat' is particularly useful when studying large numbers of paths. When studying large numbers of paths, this module will optionally write only 'crit.dat' and not writing 'paths.dat'.

### Module 5 ..... GENFMT

**Purpose of Module:** Calculate scattering amplitudes and other XAFS parameters

**Input files:** 'mod5.inp', 'global.dat', 'phase.bin', and 'paths.dat'

**Output files:** 'feff.bin', and 'list.dat'

**Other output:**

**Description:** GENFMT reads input files, and writes a file 'feff.bin' which contains all the EXAFS information for the paths, and 'list.dat' which tells you some basic information about them. These files are the main output of FEFF for EXAFS analysis. To read 'feff.bin' into your own program, use the subroutine feffdt as an example.

---

**Module 6** ..... FF2CHI

**Purpose of Module:** Calculate specified x-ray spectrum

**Input files:** 'mod6.inp', 'global.dat', 'list.dat', 'feff.bin', 'fms.bin', 'xsect.bin'

**Output files:** 'chi.dat' and 'xmu.dat'

**Other output:** 'chipNNNN.dat' and 'feffNNNN.dat'

**Description:** FF2CHI reads 'list.dat', 'fms.bin', 'feff.bin', and writes 'chi.dat' with the total XAFS from the paths specified in 'list.dat'. Additional instructions are passed to FF2CHI from 'feff.ior', so you can change S02, Debye temperature and some other parameters without re-doing the whole calculation. The file 'list.dat' can be edited by hand to change the paths being considered, and individual 'chipNNNN.dat' files with  $\chi(k)$  from each path are optionally written. If the XANES, DANES, FPRIME or XNCD card is specified, FF2CHI will write the calculated requested data in 'xmu.dat'. Various corrections are possible at this point — see input cards above.

There is an internal limit on the number of paths (set to 1200) that will be read from 'feff.bin'. This limit was chosen to handle any reasonable problem without using an excessive amount of memory. If you must use more paths, change the parameter `npv` in the FEFF source in subroutine `ff2chi` to whatever you need. It will need more memory. We have not had a case where the filter criteria were not able to solve the problem with fewer than 1200 paths.

## 3.2 Descriptions of output files

### 3.2.1 Intermediate output files

**'modN.inp' and 'ldos.inp'**

This ASCII file contains basic information from 'feff.inp' for a particular module. They still can be edited, for example to take advantage of the symmetries.

**'global.dat'**

This ASCII file contains global information about x-ray polarization and about configurational averaging.

**'geom.dat'**

This ASCII file contains Cartesian coordinates of all atoms and first-bounce information for the degeneracy reduction in pathfinder.

**'pot.bin'**

Charge density and potential (SCF or not) for all types of atoms. This file is used by XSPH module.

**'phase.bin'**

This is a binary file with the scattering phase shifts for each unique potential and with

relativistic dipole matrix elements, normalized to total cross section in `'xsect.bin'`. It is used by the FMS, pathfinder and GENFMT.

`'xsect.bin'`

Total atomic cross section for x-ray absorption. This is ASCII file, but highly sensitive to format. Information can be viewed, but the editing of this file is highly not recommended.

`'ldosNN.dat'`

$\ell$ -projected density of states for the NN<sup>th</sup> potential index (see the LDOS card)

`'fms.bin'`

contains the results of FMS calculations. Used by FF2CHI to get total XAFS or XANES.

`'paths.dat'`

Written by the pathfinder, this is a description of all the paths that fit the criteria used by the pathfinder. It is used by GENFMT. The path descriptions include Cartesian coordinates of atoms in the path, scattering angles, leg lengths and degeneracy. For details on editing this by hand, see Section 3.3. `'pathNN.dat'` files are created during the LDOS calculations for each type of potential, but they are deleted after use.

`'crit.dat'`

Values of the quantities tested against the various criteria in the pathfinder.

`'list.dat'`

List of paths to use for the final calculations. Written by GENFMT when the xafs parameters are calculated and used by FF2CHI. It contains the curved wave importance ratios, which you may wish to study. For details on editing this by hand, see Section 3.3.

The curved wave importance ratios are the importance of a particular path relative to the shortest single scattering path.

### 3.2.2 Diagnostic files from XSPH

`'misc.dat'`

Header file for quick reference.

`'phaseNN.dat'`

Complex phase shifts for each shell.

`'phminNN.dat'`

Real part of phase shifts for  $\ell=0,1,2$  only. They are smaller versions of corresponding `'phaseNN.dat'`.

`'potNN.dat'`

Detailed atomic potentials and densities.

`'atomNN.dat'`

Diagnostic information on Desclaux free atom NN.

## 3.2.3 Main output data

**‘chi.dat’**

Standard xafs data containing  $k$ ,  $\chi(k)$ ,  $|\chi(k)|$  relative to threshold ( $k = 0$ ). The header also contains enough information to specify what model was used to create this file.

**‘xmu.dat’**

The file ‘xmu.dat’ contains both XANES and XAFS data  $\mu$ ,  $\mu_0$ , and  $\tilde{\chi} = \chi\mu_0/\mu_0(\text{edge} + 50\text{eV})$  as functions of absolute energy  $E$ , relative energy  $E - E_f$  and wave number  $k$ .

**‘feff.bin’**

A binary file that contains all the information about the XAFS from all of the paths. This replaces the old ‘feffNNNN.dat’ files (which you can make using the PRINT card). If you want to use this file with your own analysis package, use the code in subroutine feffdt as an example of how to read it.

**‘feffNNNN.dat’**

You have to use the PRINT option to obtain these files. Effective scattering amplitude and phase shift data, with  $k$  referenced to threshold for shell  $nn$ :  $k$ ,  $\phi_c$ ,  $|F_{\text{eff}}|$ ,  $\phi_{\text{eff}}$ , the reduction factor,  $\lambda$ ,  $\text{Re}(p)$ .

If you need these, use the PRINT option for FF2CHI greater than 3, which will read ‘feff.bin’ and write the ‘feffNNNN.dat’ files in exactly the form you’re used to.

**‘fpf0.dat’**

Thompson scattering amplitude  $f_0(Q)$  and constant contribution to  $f'$  from total energy term.

**‘ratio.dat’**

Ratio  $\mu_0(E)$ ,  $\rho_0(E)$  and their ratio versus energy, for XMCD sum rules normalization.

## 3.2.4 Variables in the EXAFS and XANES formulae

$k$  The wave number in units of  $\text{\AA}^{-1}$ .  $k = \sqrt{E - E_f}$  where  $E$  is energy and  $E_f$  is the Fermi level computed from electron gas theory at the average interstitial charge density.

$\chi(k)$

$$\chi(k) = S_0^2 \mathcal{R} \sum_{\text{shells}} \frac{N F_{\text{eff}}}{k R^2} \exp(-2r/\lambda) \sin(2kR + \phi_{\text{eff}} + \phi_c) \exp(-2k^2 \sigma^2)$$

$\phi_c$  The total central atom phase shift,  $\phi_c = 2\delta_{\ell,c} - \ell\pi$

$F_{\text{eff}}$  The effective curved-wave backscattering amplitude in the EXAFS formula for each shell.

$\phi_{\text{eff}}$  The phase shift for each shell

$\mathcal{R}$  The total central atom loss factor,  $\mathcal{R} = \exp(-2\text{Im}(\delta_c))$

|             |                                                                                                |
|-------------|------------------------------------------------------------------------------------------------|
| $R$         | The distance to central atom for each shell                                                    |
| $N$         | The mean number atoms in each shell                                                            |
| $\sigma^2$  | The mean square fluctuation in $R$ for each shell                                              |
| $\lambda$   | The mean free path in Å, $\lambda = 1/ \text{Im } p $                                          |
| $k_f$       | The Fermi momentum at the average interstitial charge density                                  |
| $p(r)$      | The local momentum, $p^2(r) = k^2 + k_f^2(r) + \Sigma - \Sigma_f$                              |
| $\Sigma(E)$ | The energy dependent self energy at energy, $\Sigma_f$ is the self energy at the Fermi energy. |
| $\mu(E)$    | The total absorption cross-section                                                             |
| $\mu_0(E)$  | The embedded atomic background absorption                                                      |

### 3.3 Program control using intermediate output files

In addition to the CONTROL card and other options in ‘`feff.inp`’, some parameters in the files read by the various modules can be changed. For example, you can create your own paths by editing ‘`paths.dat`’ and explicit change Debye–Waller factors in the final result by editing ‘`list.dat`’.

Users may edit the some files as a quick and sometimes convenient way to prepare a given run. It is easiest to use an existing file as a template as the code which reads these files is fussy about the format of the files.

#### 3.3.1 Using ‘`paths.dat`’

You can modify a path, or even invent new ones, such as paths with more than the pathfinder maximum of 8 legs. For example, you could make a path to determine the effect of a focusing atom on a distant scatterer. Whatever index you enter for the path will be used in the filename given to the ‘`feffnnnn.dat`’ file. For example, for the choice of index 845, the EXAFS parameters will appear in ‘`feff0845.dat`’. A handy way to add a single scattering path of length  $R$  is to make a 2-leg path with the central atom at  $(0, 0, 0)$  and the scatterer at  $(R, 0, 0)$ .

GENFMT will need the positions, unique potentials, and character tags for each atom in the path. The angles and leg lengths are printed out for your information, and you can omit them when creating your own paths by hand. The label lines in the file are required (there is code that skips them, and if they’re missing, you’ll get wrong results).

#### 3.3.2 Using ‘`list.dat`’

This is the list of files that ff2chi uses to calculate chi. It includes the paths written by module GENFMT, curved wave importance factors, and user-defined Debye–Waller factors. If you

want to set Debye–Waller factors for individual paths, you may edit this file to set them. FF2CHI will sum the Debye–Waller factors in this file with the correlated Debye model  $\sigma^2$  and the global  $\sigma^2$ , if present. You may also delete paths from this file if you want to combine some particular set of paths. (CAUTION: Save the original, or you’ll have to re-run GENFMT!)

### **3.3.3 Using ‘geom.dat’**

This file can be manually edited to take advantages of the paths symmetries. See the NOGEOM card.

## Chapter 4

# Calculation Strategies and Examples

### 4.1 General comments

Self-consistent or overlapped atom potentials are necessary for the calculation of the scattering phase shifts. Self-consistent calculations take more time, and are often essential for XANES, especially for cases with significant charge transfer. Although the effect of self-consistency on EXAFS is small, such calculations give an accurate determination of  $E_0$ , thus eliminating an important parameter in EXAFS distance determinations.

Scattering phase shifts for each unique potential are necessary for FMS, PATHS and GENFMT. They are needed for the importance filters in PATHS and are the basis of the XAFS parameters calculation in GENFMT. This part of the calculation is relatively slow, so it is usually best to run it only once and use the results while studying the paths and XAFS.

To enumerate the necessary paths, the pathfinder module PATHS needs the atomic positions of any atoms from which scattering is expected. If the structure is completely unknown, only single-scattering paths can be created explicitly. Because the number of possible paths increases exponentially with total path length, one should start with a short total path length, examine the few paths (representing scattering from the nearest neighbors), and gradually increase the total path length, possibly studying the path importance coefficients and using the filters to limit the number of paths. This process is not automated, and if done carelessly can yield so many paths that no analysis will be possible.

Finally, use GENFMT to calculate the XAFS parameters, and FF2CHI to assemble the results into a chi curve. Here, the slow part is GENFMT and FF2CHI is very fast. Therefore, to explore parameters such as Debye–Waller factors, mean free path and energy zero shifts, various combinations of paths and coordination numbers, run only module FF2CHI using the results saved from GENFMT.

There are three ways to modify the Debye–Waller factor, all of which affect only the module FF2CHI. The DEBYE card calculates a Debye–Waller factor for each path. The SIG2 card adds a constant Debye–Waller factor to each path. And you can edit ‘list.dat’ to add a particular Debye–Waller factor to a particular path. These three Debye–Waller factors are

summed, so if the DEBYE and SIG2 cards are present, and if you have added a Debye–Waller factor to a particular path, the Debye–Waller factor used will be the sum of all three. See documentation below for details.

If your model changes significantly, the phase shifts (which are based in part on the structure of the material) should be recalculated. Any time the phase shifts change, the XAFS parameters will also have to be re-calculated. If the path filters have been used, the path list will also have to be recomputed.

## 4.2 EXAFS calculation

### 4.2.1 SF<sub>6</sub> Molecule

SF<sub>6</sub> Molecule. This is the simplest example of running FEFF to obtain EXAFS. Only 2 input cards are necessary. Only ‘chi.dat’ file is produced.

TITLE Molecular SF6

#### POTENTIALS

|   |      |    |     |                                     |
|---|------|----|-----|-------------------------------------|
| * | ipot | z  | tag |                                     |
|   | 0    | 16 | S   | absorbing atom must be unique pot 0 |
|   | 1    | 9  | F   |                                     |

#### ATOMS

|   |       |       |       |      |                  |
|---|-------|-------|-------|------|------------------|
| * | x     | y     | z     | ipot |                  |
|   | 0     | 0     | 0     | 0    | S absorber       |
|   | 1.56  | 0     | 0     | 1    | 6 F backscatters |
|   | 0     | 1.56  | 0     | 1    |                  |
|   | 0     | 0     | 1.56  | 1    |                  |
|   | -1.56 | 0     | 0     | 1    |                  |
|   | 0     | -1.56 | 0     | 1    |                  |
|   | 0     | 0     | -1.56 | 1    |                  |

### 4.2.2 Solids

#### Cu metal

Cu, fcc metal, 4 shells. The list of atomic coordinates (ATOMS card) for crystals can be produced by program ATOMS. Thus instead of giving long atoms list, we present a short ‘atoms.inp’ file. For connection with EXAFS fitting programs see Section 3 and also the PRINT card on page 8.

TITLE Cu crystal, 4 shells

\* Cu is fcc, lattice parameter a=3.61 (Kittel)

```
*Cu at 190K, Debye temp 315K (Ashcroft & Mermin)
DEBYE 190 315 0
```

```
POTENTIALS
  0 29 Cu0
  1 29 Cu
```

```
ATOMS
atoms list generated using atoms.inp file below
-----
title Cu metal fcc a=3.6032
fcc          ! shorthand for F M 3 M
rmax= 11.13  a=3.6032
out=feff.inp          ! index=true
geom = true
atom
!  At.type    x    y    z
Cu      0.0  0.0  0.0
-----
```

### YBCO High-Tc superconductor

```
TITLE YBCO: Y Ba2 Cu3 07      Cu2 core hole
```

```
CONTROL 1 1 1 1 1 1
PRINT   0 0 0 0 0 0
```

```
RPATH 4.5
```

```
POTENTIALS
*  ipot  z  tag
   0  29  Cu2
   1   8   0
   2  39   Y
   3  29  Cu1
   4  56   Ba
```

```
ATOMS
atoms list generated by the following atoms.inp file
-----
title YBCO: Y Ba2 Cu3 07 (1-2-3 structure)
space P M M M
rmax=5.2          a=3.823 b=3.886 c=11.681
```

```

core = Cu1
atom
! At.type  x      y      z      tag
  Y        0.5    0.5    0.5
  Ba        0.5    0.5    0.184
  Cu         0      0      0      Cu1
  Cu         0      0    0.356    Cu2
  O          0      0.5    0      O1
  O          0      0    0.158    O2
  O          0      0.5    0.379    O3
  O          0.5    0    0.377    O4
-----

```

### 4.2.3 Estimate of $S_0^2$

All above examples yield calculations for K edge (default). To do calculations for other edges use EDGE (or HOLE) cards. These cards will also yield an estimate of  $S_0^2$  from atomic calculations if you set S02 = 0 by the one of three possible ways shown below.

```

EDGE    L3    0.0
HOLE     4    0.0
S02      0.0

```

The result for S02 is given in 'chi.dat' or 'xmu.dat' files.  $S_0^2$  is a square of determinant of overlap integrals for core orbitals calculated with and without core hole. The core-valence separation can be changed by editing subroutine `getorb`, but it is currently set by default to the most chemically reasonable one.

### 4.2.4 Configuration averaging over absorbers

In amorphous materials or materials with distortions from regular crystals the absorbing atoms (with the same number in periodic table) may have different surroundings. Thus one may want to average the calculation over different types of sites for the same atom or even over all atoms in the 'feff.inp' file. This can be accomplished using CFAVERAGE card of Section 2.2.

### 4.2.5 Adding self-consistency

Self-consistency is expected to be more important for the XANES calculations, but even for EXAFS one may want to have more reliable determination of Fermi level or to account for the charge transfers in order to do fits with single energy shift  $E_0$ . Our experience shows that reliable EXAFS phase shifts are best achieved by using SCF card.

```

SCF      3.8

```

The above example works for solids or large molecules, but for molecules with less than 30 atoms calculations can be done faster if you set `lfms1 = 1`,

```
SCF    10.0  1
```

For details see SCF and FMS cards in Sections 2.3 and 2.5.

## 4.3 XANES calculations

### 4.3.1 Need for SCF and additional difficulties for XANES

XANES calculations are usually more challenging than EXAFS calculations. They usually take more time and require more experience from the user. Fortunately FEFF8 automates many steps in the procedure and includes important full multiple scattering terms and self-consistency. This improves on the high order path expansion approach to XANES in FEFF7 which only allowed the maximum number of paths of amplitude larger than a plane wave criteria in PCRITERIA card. To account for the poor electron gas estimates of the Fermi level, CORRECTIONS cards were needed. Moreover, to obtain good results for the spectra, one had also to play with AFOLP, EXCHANGE and ION cards, which alter the way the scattering potential is constructed in somewhat uncontrolled ways.

One of the main advantages of FEFF8, is that it yields self-consistent potentials using the SCF card. The use of the SCF card also gives a more reliable estimate of Fermi level (the CORRECTIONS card can still be used, since the error in Fermi level position is only a few eV). FEFF8 thus automatically accounts for charge transfer. The ION card should be used only to specify the total charge of a cluster. AFOLP in general leads to better results for XANES and is done by default. With FEFF7 we also had to use EXCHANGE 5 model for Pu hydrates, but with the self-consistent FEFF8 the standard EXCHANGE 0 works well. Thus the use of self-consistency leads to closer results for different exchange correlation models.

The use of the high order MS path expansion and PCRITERIA can lead to unreliable XANES calculations when the MS series converges poorly (for example, near the Fermi level). Thus the inclusion of FMS capabilities in FEFF8 is essential for calculations of LDOS and electronic densities and is often an improvement on path calculations of XANES. Actually we suggest that for LDOS calculations one uses FMS exclusively and uses path expansion for testing its convergence. This will cost CPU time, but will lead to more reliable XANES results. The FMS calculations for a cluster of 87 atoms typically take more time and memory than the other 5 modules. The results can be somewhat better with larger clusters, but typically one achieves convergence with about 50-200 atoms and calculational time scales as a third power of number of atoms in a cluster and quickly becomes prohibitive. Below we present several sample input files for XANES calculations.

4.3.2 GeCl<sub>4</sub> Molecule

This is a historically first molecule for which EXAFS was calculated by Hartree, Kronig and Peterson (1934), using the short range order theory.

TITLE    GeCl\_4    r=2.09 /AA

NOHOLE

HOLE 1    1.0

RSIGMA

CONTROL    1    1    1    1    1    1

SCF        3.0    1

FMS        3.0    1

RPATH    1.0

XANES    8.0 0.05

AFOLP    1.30

POTENTIALS

|   |      |    |        |
|---|------|----|--------|
| * | ipot | z  | label  |
|   | 0    | 32 | Ge 3 3 |
|   | 1    | 17 | Cl 3 3 |

ATOMS

|   |         |         |         |      |      |          |
|---|---------|---------|---------|------|------|----------|
| * | x       | y       | z       | ipot | atom | distance |
|   | 0.0000  | 0.0000  | 0.0000  | 0    | Ge   |          |
|   | 1.2100  | 1.2100  | 1.2100  | 1    | Cl   |          |
|   | 1.2100  | -1.2100 | -1.2100 | 1    | Cl   |          |
|   | -1.2100 | 1.2100  | -1.2100 | 1    | Cl   |          |
|   | -1.2100 | -1.2100 | 1.2100  | 1    | Cl   |          |

END

## 4.3.3 Solid: XANES and LDOS

BN crystal has a zinc sulfide structure, and is a case in which a multiple scattering expansion does not converge near the Fermi level. Using the full multiple scattering approach leads to good agreement with experiment.

TITLE    BN cubic zinc sulfide structure

CONTROL    1 1 1 1 1 1

PRINT        5 0 0 0 0 0

SCF    3.1

```

HOLE 1 1.0 1=k edge, s0^2=1.0
EXCHANGE 0 0 1.0
LDOS -20 10 0.5
FMS 5.1
RPATH 1.0
XANES 4.0

```

```

INTERSTITIAL 0 1.54

```

#### POTENTIALS

```

* ipot z label lmax1 lmax2
    0 5 B 2 2 0.1
    1 7 N 2 2 1
    2 5 B 2 2 1

```

#### ATOMS

```

list generated by ATOMS program

```

```

-----
title BN (zincblende structure)
Space zns
a=3.615 rmax=8.0 core=B
atom
! At.type x y z tag
  B 0.0 0.0 0.0
  N 0.25 0.25 0.25
-----

```

### 4.3.4 Absolute cross-section

The absolute cross section can be obtained from the output in 'xmu.dat'. Look for this line:

```

xsedge+100, used to normalize mu          2.5908E-04

```

Since our distances are in Å, we report cross section also in Å<sup>2</sup>. If you multiply the 4-th or 5-th column by this normalization value you will obtain the cross section in Å<sup>2</sup>. Often the absolute cross section are reported in barns, which has simple connection with our units (1 Å<sup>2</sup> = 100Mbarn)

## 4.4 Spin-dependent calculations

### 4.4.1 General description

These calculations have been automated in FEFF8.40. For all spin-dependent calculations, the SPIN card must be added to 'feff.inp'. For the calculations to be automatic, FEFF must be

compiled with `nspix = 2` (found in `'feff84/src/HEADERS/dim.h'`). See SPIN card for additional details.

This section contains instructions for extracting the XMCD signal and a description of the SPXAS technique, as well as example input files for calculations of SPXAS and XMCD.

In the case that the code has been compiled with `nspix = 1`, the output will not contain the final result and the code must be run twice, once for spin up and down. An additional simple program `'spin.f'` is useful for taking care of the different normalizations and giving the finite results in XMCD or SPXAS (4.4.3).

The same paths should be used for spin up and down calculations, otherwise the difference between 2 calculations may be due to different paths used. Typically the paths list in `'paths.dat'` should be generated by running the usual EXAFS calculations and comparing with experiment (to make sure that all important paths are included). Then when running with SPIN turn off the pathfinder module using CONTROL card. This is probably the only time you have to skip the pathfinder module.

`'spin.f'`:

```

implicit double precision (a-h,o-z)
c      This program read two xmu.dat files for spin -up and -down,
c      calculated with Feff8.20 for the SAME paths list.
c      spin-up file is fort.1, spin-down file is fort.2
c      Both have to be edited: All lines should be deleted except
c      1) line: xsedge+100, used to normalize mu          1.3953E-04
c      leave only on this line:  1.3953E-04
c      2) 6-column data lines
c      The output will be written in fort.3 in 6 columns
c      E+shift1 E(edge)+shift2  xk cmd_total cmd_background  cmd_fs
c      where total = atomic background + fine structure

c      There are 3 possibilities
c      case 1) you want XMCD signal and used SPIN \pm 1
c      case 2) you want XMCD signal and used SPIN \pm 2, in order
c      to use non-relativistic formula for XMCD
c      factor  $li/2j+1$  which was not convenient to do in a program
c      case 3) you want SPXAFS and used SPIN \pm 2
c      ENTER your case here (icase is positive integer only)
c      icase = 2

c      if icase=2 ENTER factor= $(-1)^{L+1/2-J} * L/(2J+1)$ 
c      where L,J are for your edge (ex. for L3 L=1 J=3/2, for L2 L=1 J=1/2)
c      for L3
c      factor = 0.25
c      for L2

```

---

```

c      factor = -0.5

c      ENTER the energy shift you want for columns 1 and 2 in xmu.dat
      shift1 = 0
      shift2 = 0

c      everything below is automated further
      read (1,*,end=10) ap
      read (2,*,end=10) am
      xnorm = 0.5 *(ap+am)
c      read the data
3      read(1,*,end=10)  x1, x2, ek, y1, y2, y3
      read(2,*,end=10)  x1, x2, ek, z1, z2, z3
      if (icase.eq.1) then
c          no xafs in this case:xfs - atomic part of XMCD
          t1 = (y1*ap + z1*am)/xnorm
          t2 = (y2*ap + z2*am)/xnorm
          t3 = (y3*ap + z3*am) /xnorm

          elseif (icase.eq.2) then
              t1 = (y1*ap - z1*am)*factor /xnorm
              t2 = (y2*ap - z2*am)*factor /xnorm
              t3 = (y3*ap - z3*am)*factor /xnorm

          elseif (icase.eq.3) then
c              factor=0.5 always for SPXAFS
              t1 = (y1*ap - z1*am)/2.0/xnorm
              t2 = (y2*ap - z2*am)/2.0/xnorm
              t3 = (y3*ap - z3*am)/2.0/xnorm
c              you may want average total XAS as output in last column
c              t3 = (y1*ap + z1*am)/2.0/xnorm
          endif
          x1 =x1 + shift1
          x2 =x2 + shift2
          write(3,5)  x1, x2, ek, t1, t2, t3
5      format (6e13.5)
      goto 3
10 continue
      stop
      end

```

## 4.4.2 XMCD - Gd L1 edge

Example input file for calculation of XMCD for the Gd L1 edge. Note the presence of both SPIN and XMCD cards.

```

TITLE    Gd  l1  hcp

HOLE 2    1.0      2=l1 edge, s0^2=1.0
SPIN     1
EXCHANGE 2    0.0  0.0

CONTROL   1      1      1      1      1      1

RPATH     7.29
PRINT     5 0 0 0 0 3

CRITERIA  0.0  0.0  curved  plane
DEBYE     150   176  temp    debye-temp
XANES
XMCD

POTENTIALS
*   ipot   z   label
      0    64   Gd
      1    64   Gd

ATOMS
the list of atoms is created by ATOMS program
-----
title    Gd , hcp
! Wyckoff, vol.1 p.331
space    hcp
rmax = 9.0
a = 3.6354
c = 5.7817
atom
  Gd    0.33333  0.66667  0.25  center
-----

```

## 4.4.3 SPXAS

For antiferromagnets, a measure of the spin-up and spin-down signals can be accomplished using SPXAS. SPXAS is a technique where you measure the spin-up and spin-down signal by measuring intensity of two spin-split  $K_\beta$  lines. This corresponds to measuring spin-order

relative to the spin on the absorber (not relative to the external magnetic field as in XMCD). As an example, let us look at the Mn K edge of antiferromagnetic  $\text{MnF}_2$ . Our calculations agree well with experiment in EXAFS region.

Here is an example input file for  $\text{MnF}_2$  (SPXAS)

```

TITLE    MnF2 (rutile) cassiterite (Wykoff)

HOLE 1    1.0      1=k edge, s0^2=1.0
SPIN    -2

CONTROL   1        1        1        1        1        1
PRINT     0         0         0         2         0         4
EXCHANGE  0  0.0  0.0
CORRECTIONS 0.0  0.0

RPATH    10.0
XANES

PCRITERIA      0.8      40.0
*CRITERIA      curved   plane
CRITERIA       0.0      0.0
*DEBYE         temp    debye-temp
DEBYE          300      350
NLEG           4

POTENTIALS
*   ipot   z   label
      0   25   Mnup
      1    9    F
      2   25   Mnup
      3   25   Mndown

ATOMS
  0.0000    0.0000    0.0000    0   Mnup    0.0000
  1.4864    1.4864    0.0000    1    F      2.1021
 -1.4864   -1.4864    0.0000    1    F      2.1021
  0.9503   -0.9503    1.6550    1    F      2.1319
  0.9503   -0.9503   -1.6550    1    F      2.1319
 -0.9503    0.9503    1.6550    1    F      2.1319
 -0.9503    0.9503   -1.6550    1    F      2.1319
  0.0000    0.0000   -3.3099    3   Mndown   3.3099
  0.0000    0.0000    3.3099    3   Mndown   3.3099
 -3.3870    1.4864    0.0000    1    F      3.6988
  3.3870   -1.4864    0.0000    1    F      3.6988

```

---

|         |         |         |   |        |        |
|---------|---------|---------|---|--------|--------|
| 1.4864  | -3.3870 | 0.0000  | 1 | F      | 3.6988 |
| -1.4864 | 3.3870  | 0.0000  | 1 | F      | 3.6988 |
| 2.4367  | 2.4367  | -1.6550 | 2 | Mnup   | 3.8228 |
| -2.4367 | -2.4367 | -1.6550 | 2 | Mnup   | 3.8228 |
| 2.4367  | -2.4367 | -1.6550 | 3 | Mndown | 3.8228 |
| -2.4367 | -2.4367 | 1.6550  | 3 | Mndown | 3.8228 |
| -2.4367 | 2.4367  | -1.6550 | 3 | Mndown | 3.8228 |
| 2.4367  | 2.4367  | 1.6550  | 3 | Mndown | 3.8228 |
| 2.4367  | -2.4367 | 1.6550  | 2 | Mnup   | 3.8228 |
| -2.4367 | 2.4367  | 1.6550  | 2 | Mnup   | 3.8228 |
| ...     |         |         |   |        |        |
| END     |         |         |   |        |        |

## 4.5 Elastic Scattering Amplitudes

All necessary components to obtain the elastic scattering amplitude can be calculated with the FEFF8.2 code and above. Thus Thomson scattering amplitudes are written in file ‘**fpf0.dat**’, elastic amplitude near some specific edge are calculated with DANES card while far from the edge with FPRIME card which neglects solid state effects on  $f'$ . The  $f''$  can be obtained with XANES card. The formula connecting  $f''$  and the absorption cross section  $\sigma$  is (in atomic units)  $f'' = \omega c \sigma / 4 / \pi$ . For calculations at energies well above the absorption edge we found that ground state potentials yields better results, and also that quadrupolar transitions have to be included.

## 4.6 X-ray Emission Spectra XES

The nonresonant x-ray emission spectra (fluorescence spectra) are treated in the same way as the x-ray absorption process for states below the Fermi level. To perform these calculations one simply replaces the XANES card with XES. Preliminary comparisons with experiment for phosphorus  $K_\beta$  line show good agreement with experiment for various compounds. Further tests are in progress. Please report any problems with this card to the authors.

## 4.7 Local Field and Core-hole Effects (PMBSE, TDDFT)

These are features of FEFF that are still under development, and not reliable in this release.

Many-body effects such as local fields and the core-hole interaction can be significant in x-ray absorption spectra even several hundred eV above an absorption edge. The treatment of these effects requires theories beyond the independent-particle approximation, e.g., the Bethe-Salpeter equation (BSE) or the time-dependent density-functional theory (TDDFT).

The projection-operator method Bethe-Salpeter equation (PMBSE) is used for core-hole absorption spectra calculations. The BSE is usually limited to low energies, while the TDDFT often ignores the nonlocality of the core-hole interaction. Time dependent density functional theory (TDDFT) is a general framework for studying non-stationary electronic processes. TDDFT is used for local-field absorption spectra calculations.

Here we present a combined approach for the calculations of the x-ray spectra that include both of these effects, together with inelastic losses and self-energy shifts over a wide energy range.

Note that the TDLDA card accounts for some of the same effects, and is reliable in this release of FEFF8.

See *Combined Bethe-Salpeter equations and time-dependent density-functional theory approach for x-ray absorption calculations*; A. L. Ankudinov, Y. Takimoto, and J. J. Rehr, Phys. Rev. B 71, 165110 (2005)

## Appendix A

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### A.1 Restrictions and license information

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FEFF8: A.L. Ankudinov, B. Ravel, J.J. Rehr, and S.D. Conradson, Phys. Rev. B 58, pp. 7565-7576 (1998).

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## Appendix B

# Installation Instructions

The program FEFF8.20 is provided as a single file ‘feff82.f’. An experimental modular form consisting of eight source files ‘rdinp-tot.f’, ‘pot-tot.f’, ‘ldos-tot.f’, ‘xsph-tot.f’, ‘fms-tot.f’, ‘path-tot.f’, ‘genfmt-tot.f’ and ‘ff2x-tot.f’ is available by request. LDOS calculations are still executed in second module XSPH, even though it is now separate program. This is done for compatibility with other feff8 input files. The modular code has requires less memory and yields more effective compilation and execution on small machines; however, the CFAVERAGE card is disabled in the modular version. Parallel execution on MPI clusters has been tested for modular code only and is still experimental. For details, please contact the authors.

Each file contains a main program and all necessary subroutines. Simply compile (and link) every program on your system using a Fortran 77 compiler (and your usual linker), e.g., for any UNIX system the command `f77 -o module module.f` is usually sufficient, but may be augmented by optimization flags.

For the monolithic version, there is only a single executable to run. For the modular code, the 8 executable files must be run in the order listed: `rdinp`, `pot`, `ldos`, `xsph`, `fms`, `path`, `genfmt`, `ff2x`, so that the necessary input files for each successive module are produced. We suggest that you write a small script (named `feff8`), that executes all modules on your OS, if you use the modular code. Samples are available from the authors. Parallel versions of the modules and appropriate compile and run scripts are also available and require a working version of the MPI libraries. Please contact the authors for details.

Typically the monolithic FEFF8 requires about 64 MB of RAM to run for clusters of up to 87 atoms. To adjust the required memory to your computer, you should change the `nclusx` parameter *globally*. The required memory scales approximately as  $nclusx^2$ . For larger clusters, you may wish to reduce the maximum angular momentum `lx` to keep the total size reasonable, e.g., set `nclusx = 300`, `lx = 2` globally in all subroutines.

FEFF8 will also run on older ‘feff.inp’ files for FEFF7 (and also FEFF6) and will yield results for EXAFS and XANES comparable to older versions of FEFF. The first module of FEFF7 has now been split into 3 modules and therefore FEFF8 will use the first value in FEFF7 CONTROL and PRINT cards for the first 3 modules of FEFF8. But we suggest that users make

use of the SCF, FMS, LDOS and other cards to take advantage of FEF8's new capabilities.

The FEF8 code is written in ANSI Fortran 77, except that `complex*16` variables are required. Since data type `complex*16` is not part of the ANSI standard, minor compiler-dependent modification may be necessary. We have used the VAX extensions to ANSI Fortran 77 since they seem to be the most portable. The non-standard statements and intrinsic functions used are: `complex*16` variables and arrays; `dimag(arg)` returns a double precision imaginary part of its argument; `dcmplx(arg)` returns a `complex*16` version of its argument; `DBLE(arg)` returns a double precision real part of its argument; `sqrt`, `exp`, `abs` and other generic math functions are assumed to accept `complex*16` arguments and return double precision or `complex*16` results.

Files are opened with the `open` statement. As the ANSI definition gives some leeway in how the `open` statement interacts with the operating system, we have chosen file names and conventions that work on UNIX, VAX/VMS, IBM PCs (and clones) with MS FORTRAN, CRAY, MAC's and CDC machines. It may be necessary to modify the open statements on other systems.

## B.1 UNIX

We have endeavored to make FEF8 portable to all UNIX (including HP, AIX, LINUX, Alpha, BSD and CRAY) machines without any modification. If your machine does not reproduce the test output files 'xmu.dat' and/or 'chi.dat' to high accuracy, please let us know. Also, please report any compiler problems or warning messages to the authors, as this will help us achieve full portability.

On SGI machines a significant increase in speed may be achieved with the following optimization flags

```
f77 -Ofast -LNO:opt=0 -IPA:INLINE=OFF module.f -o module
```

Inlining is a potentially dangerous operation which modifies the source code, so the output ('xmu.dat') should be checked. Switching off inlining may slow down the code, as we found for the FMS module, but some other modules would not operate properly with inlining.

On many UNIX systems it may be necessary to increase the memory stack size to allow calculations on large clusters. For example, for 200 atom clusters, you should execute the following command before running FEF8 for the CSH or one of its derivatives

```
limit stacksize 170000
```

or, if you use the Bourne shell or one of its derivatives, the following

```
ulimit -s 170000
```

If you use the GNU g77 compiler, for example, on LINUX, BSD, and other systems try for example:

```
g77 -O2 -ffast-math -m486 -Wall -g -fno-silent
```

or (since the compile flags often change ) simply

```
g77 -O2
```

Do not use the *-pedantic* flag, although a casual reading of the *g77* document would suggest it is a good idea. *-pedantic* does not allow the use of double complex which is essential in FEFF.

On some AIX machines, the intrinsic real functions need to be promoted to double precision (e.g. *-qautodbl=dbl* option on RS6000).

## B.2 CRAY, SGI-CRAY, and CDC UNIX

For CRAY, SGI-CRAY and CDC UNIX, please keep in mind the following points: Floating point calculations in FEFF are usually done to 64 bit precision. Thus for 32 bit word machines, the code uses double precision variables throughout, i.e., *real\*8* for real numbers and *complex\*16* for complex numbers. If your machine uses 8 bytes (64 bits) for single precision floating point numbers and integers (for example, CRAYs and some CDC machines), you should use the CRAY FORTRAN compiler option to ignore double precision statements in the code. The compile flags depend on machine vintage (see the *cf77* or *f90* man pages for details) and are of the form:

```
f90 -c -dp or cf77 -c -wf"-dp"
```

## B.3 MS-DOS, WIN-NT, 9X, ME, 2K, etc

Because of the awkwardness of DOS, many users do not have FORTRAN compilers and many of those compilers are difficult to use with large codes. Thus we have made executable versions of FEFF for PCs available. You will need an 486 with a math coprocessor or pentium chip and at least 64 MB of RAM in addition to that needed for DOS and any resident utilities. Further details are supplied with the executables.

If you prefer your own compiler (e.g., Compaq Visual Fortran), or are using an operating system other than DOS, simply compile the source code using your FORTRAN compiler and linker as you would for any other machine.

## B.4 Macintosh

FEFF is often difficult to compile on Macintosh machines with Mac OS 9 and below, but executable versions of FEFF for Macintosh computers including the G4 are available from the FEFF Project.

For Mac OS X UNIX use the following options with Absoft's *f77* compiler

```
f77 -N113 -N11 -f -O module.f
```

*-f* makes F77 case insensitive. *-N113* promotes intrinsic routines from real to double precision. *-N11* allows 32 bit operations, and *-O* makes basic optimizations.

There is another option for Mac OS X UNIX. At least one user has found that f2c and the cc compiler can be made to work.

## B.5 Other Machines: VMS, NEXT, etc

To compile FEFF8 on VMS (6.0 to 6.2 versions) machine we had to increase virtual memory using SYSGEN from a SYSTEM account.

```
$mc sysgen
sysgen> use current
sysgen>show virtualpagecnt
sysgen>set virtualpagecnt (value + 25%)
sysgen>write current
sysgen>exit
$reboot
```

If the code still does not compile one should further increase virtual memory. It may be impossible to compile the full code on old VAX stations. To run FEFF8 on VMS machines, you may need to increase quotas (*pgflquo=500,000*) for the users using AUTHORIZE from a SYSTEM account.

```
$run authorize
authorize> modify username /pgflquo=500000
authorize>exit
$reboot
```

Use the following options with Absoft's F77 for NeXT

```
f77 -N53 -f -s -O module.f
```

*-f* makes F77 case insensitive. *-s* makes a code for units larger than 512K. *-N53* uses the 68030/68040 processors with 68881/2 math coprocessor. *-O* is an optimization flag.

## Appendix C

## References

Please cite at least one of the following articles if FEFF is used in published work.

FEFF8 *Main FEFF8 reference* A.L. Ankudinov, B. Ravel, J.J. Rehr, and S.D. Conradson, *Real Space Multiple Scattering Calculation of XANES*, Phys. Rev. B **58**, 7565 (1998).

FEFF8.1

A.L. Ankudinov, and J.J. Rehr, *Theory of solid state contributions to the x-ray elastic scattering amplitude*, Phys. Rev. B **62**, 2437 (2000).

FEFF8.2

A.L. Ankudinov, C. Bouldin, J.J. Rehr, J. Sims, H. Hung, *Parallel calculation of electron multiple scattering using Lanczos algorithms*, Phys. Rev. B **65**, 104107 (2002).

FEFF8.4

A.L. Ankudinov, A.I. Nesvizhskii, and J.J. Rehr, *Dynamic screening effects in x-ray absorption spectra*, Phys. Rev. B **67**, 115120 (2003); this contains information about TDLDA calculations in FEFF.

FEFF7 A.L. Ankudinov and J.J. Rehr, *Relativistic Spin-dependent X-ray Absorption Theory*, Phys. Rev. B **56**, R1712 (1997). A.L. Ankudinov, PhD Thesis, *Relativistic Spin-dependent X-ray Absorption Theory*, University of Washington, (1996); this contains a review of x-ray absorption theory, a whole chapter of information about FEFF for expert users, example applications, and the full FEFF7 program tree.

FEFF6 S.I. Zabinsky, J.J. Rehr, A. Ankudinov, R.C. Albers and M.J. Eller, *Multiple Scattering Calculations of X-ray Absorption Spectra*, Phys. Rev. B **52**, 2995 (1995).

FEFF5 J.J. Rehr, S.I. Zabinsky and R.C. Albers, *High-order multiple scattering calculations of x-ray-absorption fine structure*, Phys. Rev. Lett. **69**, 3397 (1992).

FEFF3 and FEFF4

J. Mustre de Leon, J.J. Rehr, S.I. Zabinsky, and R.C. Albers, *Ab initio curved-wave x-ray-absorption fine structure*, Phys. Rev. B **44**, 4146 (1991).

FEFF3 J.J. Rehr, J. Mustre de Leon, S.I. Zabinsky, and R.C. Albers, *Theoretical X-ray Absorption Fine Structure Standards*, J. Am. Chem. Soc. **113**, 5135 (1991).

FEFF Review

J.J. Rehr and R.C. Albers, *Modern Theory of XAFS*, Rev. Mod. Phys. **72**, 621 (2000).

Sum rule normalization procedure in FEFF8.2

A.I. Nesvizhskii, A.L. Ankudinov, and J.J. Rehr, *Normalization and convergence of x-ray absorption sum rules*, Phys. Rev. B **63**, 094412 (2001).

Multiple Scattering theory in FEFF

J.J. Rehr and R.C. Albers, *Scattering-matrix formulation of curved-wave multiple-scattering theory: Application to x-ray-absorption fine structure*, Phys. Rev. B **41**, 8139 (1990).

Dirac-Fock atom code

A.L. Ankudinov, S.I. Zabinsky and J.J. Rehr, *Single configuration Dirac-Fock atom code*, Comp. Phys. Comm. **98**, 359 (1996).

## Appendix D

# Code Variables and Dimensions

The array names in FEFF are a bit cryptic due to the six character limit in standard FORTRAN — the comments given in the source code and explain what the names mean. If you need to run larger problems than the dimension statements in the code allow, simply change the dimensions in all the relevant parameter statements and recompile. The main parameters to change are `nclusx`, which specifies the maximum cluster size for full multiple scattering and `lx` which specifies the maximum angular momentum in SCF potentials. These and other user changeable parameters are listed in ‘`dim.h`’, which is now incorporated explicitly in the code. If you need help to modify the FEFF code, please contact the authors.

```

c      header file dim.h
c      caution: changing parameters other than nclusx and lx may
c      break the code; when in doubt contact the authors
c
c      maximum number of atoms for FMS. Reduce nclusx if you need
c      smaller executable.
c      parameter (nclusx=87)
c      max number of spins: 1 for spin average; 2 for spin-dep
c      parameter (nspx=1)
c      max number of atoms in problem for the pathfinder
c      parameter (natx =1000)
c      max number of atoms in problem for the rdinp and ffsort
c      parameter (nattx =10000)
c      max orbital momentum for FMS module.
c      parameter (lx=3)
c      max number of unique potentials (potph)
c      parameter (nphx = 7)
c      max number of ang mom (arrays 1:ltot+1)
c      parameter (ltot = 24)
c      Loucks r grid used through overlap and in phase work arrays

```

---

```
parameter (nrptx = 1251)
c      Number of energy points genfmt, etc.
parameter (nex = 150)
c      Max number of distinct lambda's for genfmt
c      15 handles iord 2 and exact ss
parameter (lamtot=15)
c      vary mmax and nmax independently
parameter (mtot=4, ntot=2)
c      max number of path atoms, used in path finder, NOT in genfmt
parameter (npatx = 8)
c      matches path finder, used in GENFMT
parameter (legtot=npatx+1)
c      max number of overlap shells (OVERLAP card)
parameter (novrx=8)
c      max number of header lines
parameter (nheadx=30)
```

## Appendix E

# Changes From Previous Versions of FEFF

FEFF8.40 automates spin dependent calculations, simplifying calculation of spin-polarized absorption spectra (SPXAS, SPEXAFS) and XMCD, and introduces TDLDA calculations to account for screening effects. Experimental calculations have been added using TDDFT and PMBSE to account for additional core-hole effects, but these are unreliable in this release.

FEFF8.20 extends the calculation of quadrupolar transitions and x-ray emission spectra calculations, and also permits faster XANES calculations using iterative Lanczos FMS algorithms. Improved potentials for *f*-electron materials are also included. The code has been restructured to simplify future developments and is also available as separate modules which can be run on parallel machines.

FEFF8.10 fixes a few bugs (notably that for polarization dependent calculations for initial states other than *s*-character; i.e., L2, L3 edges, etc) and adds some new capabilities. The code has been extended to calculate elastic scattering amplitude and x-ray natural dichroism. An additional output file has been added '`ratio.dat`' for use in our procedure for sum-rule normalization.

FEFF8 potentials can now be calculated self-consistently (SCF card) which also gives a more accurate Fermi level position and accounts for the charge transfer. Full multiple scattering capability was also added (FMS card). This is essential for SCF potentials, L-projected density of states (LDOS also is a new card) and often XANES. All these new cards are not essential for EXAFS calculations, but the SCF potential can be used to reduce number of EXAFS fitting parameters by calculating the Fermi level and non-integral charge counts on each site. The possibility of calculating multiple-scattering Debye-Waller factors from force constants and/or dynamical matrices has also been added. The possibility of configurational averages (CFAVERAGE card) of EXAFS (or XANES) over different absorbers of the same type has been added. The CONTROL structure has been changed to accommodate the new cards, but backward compatibility has been maintained. Also several cards (e.g. EXAFS, XANES and POTENTIALS) now have additional optional fields.

## Appendix F

# Trouble-Shooting FEFF Problems and Bug Reports

FEFF8 has been extensively tested on many different architectures, but occasionally new bugs show up. In an effort to maintain portable and trouble-free codes we take all bug reports seriously. Please let us know if you encounter any compilation error or warning messages. Often we receive reports by users of older versions of FEFF of bugs that have been fixed in more recent releases. Other code failures can often be traced to input file errors, sometimes quite subtle, and some are compiler bugs, for which we try to find a workaround.

To report a bug, please tell us the version of the code you are using and which operating system and compiler you have. Please include a `'feff.inp'` if the problem occurs after compilation and enough detail concerning the warning or error messages or other difficulties you have so that we can attempt to reproduce the problem.

Some known and commonly encountered difficulties are:

- Non-physical, widely spaced distributions of atoms. Symptoms of this common problem are very large muffin-tin radii (see the header of any `'dat'` file) and possibly a failure of the phase-shift program to converge. This gives error message `hard test fails in fovrg`.
- An error in assigning potential indices; the first atom with a given potential index must have the geometry representative of this potential type. This is sometimes fixed by using a somewhat larger cluster; in fact it is usually desirable to have a larger cluster for potential construction than that used in the XAFS calculation due to errors in the potentials at surfaces. Unless the atom distribution is physically possible, you can expect the code to have problems.
- Hash collision in the pathfinder. This is now rare, but can usually be corrected simply by changing distances in the fourth decimal place.
- For the  $M_{IV}$  and higher edges you may receive the error message like: `Lambda array overfilled`. The calculations should be repeated with `IORDER -70202` card.